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VASYUTINSKIY, N.A.; VASYUTINSKAYA, L.I. Arsenic affection in the process of firing of Kethertek iron ores. Zhur. prikl. khim. 37 no.8:1681-1689 Ag 164. (MJPA 17:1])

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GREBNEV, S.K.; VASYUTINSKIY, N.A.; VASYUTINSKAYA, L.I.

Nature of ferromagnetic oxidation products of siderites. Zhur.prikl.khim. 34 no.8:1690-1695 Ag '61. (MIRA 14:8) (Siderite) (Magnetite)

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SAVITSKIY, I.V., VASYUTINSKAYA, TO.M.

Effect of aniline on certain indicators of protein metabolism [with summary in English]. Vop.med.khim. 4 no.5:351-358 S-0 '58 (MIRA 11:11) 1. Nauchno-issledovatel'skaya sanitarno-khimicheskaya laboratoriya Ministerstvo zdravookhraneniya Ukrainskoy SSR, Odessa. (BLOOD PROTEINS, effect of drugs on, aniline (Rus)) (ANILINE DYES, effects, on blood proteins (Rus))

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SAV	ITSKIY, I.V. [Savyts'kyi, I.V.]; VASYUTINSKAYA, Ye.M. [Vasiutyns'ka, IB.M.]		
	Studying certain aspects of protein metabolism during aniline intoxication and tissue therapy. Fiziol.zhur. 6 no.1:101-107 Ja-F 460. (MIRA 13:5)		
	<pre>l. Odesskaya nauchno-issledovatel'skaya sanitarno-khimicheskaya laboratoriya. (PROTEIN METABOLISM) (TISSUE EXTRACTS) (ANILINEPHYSIOLOGICAL EFFECT)</pre>		

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L 19785-65 ACCESSION NR: AR4045761

Ometabolism were investigated. In the first days following transplantation, the level of serum protoins was reduced due to albumin and the globulin (gamma globulin) level increased. At later stages the globulin level decreased, particularly the algoar and beta-frictions, and the cuncentration of restoration of the labor was related auf the connects and the solid of the solid sector and the sector and the solid sector and the solid sector and the solid sector sector and the solid sector uf untro uncharred akir flaga and graavryet arit flags. The arts ra This that the conditions is the second state of the second state of the second state of the second state of the and the state of the

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VASYUTINSKIY, A. I.

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Vasyutinskiy, A. I. -- "Study of the Influence of Substituents on Certain Analytical Properties of Bensoic Acids and the Working Out on This Basis of Methods of Quantitative Determination of Inorganic Preparations." Tartu State U, Tartu, 1955 (Dissertation for the Degree of Candidate of Pharmaceutical Sciences.)

SO: Knizhnaya Letopis', No. 24, Moscow, Jun 55, pp 91-104

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AUTHORS:	Portnov, A. I., Vasyutinskiy, A. I. 75-13-3-11/27
TITLE:	The Use of 2-Sodium- Paraaminobenzoic Acid for the Titrimetric Determination of Cadmium (Primemeniye paraaminobenzoata matriya dlya titrimetricheskogo opredeleniya kadmiya)
PERIODICAL:	Zhurnal analiticheskoy khimii, 1958, Vol 13, Nr 3, pp. 319 - 322 (USSR)
ABSTRACT:	Many methods for the quantitative determination of cadmium have been described in publications (Refs 1-7). Complexometric in importance. When research was undertaken into the influence of substitutes on the solubility of the salts of benzoic acid, with 2-sodium-aminobenzoic acid. This observation led to the the cadmium. This method is based on the precipitation of cad- nium by solution of 2-sodium-aminobenzoate and further bromato- in the precipitate. An investigation of the caminobenzoic acid uncertained of the remaining para-aminobenzoic acid
Card 1/3	in the precipitate. An investigation of the composition of the precipitate resulted in the following formula:
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The Use of 2-Sodium-Paraaminobenzoic Acid for the Titrimetric Determination of Cadmium

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 $(H_2NC_6H_4COO)_2Cd.3H_2O.$ Tests showed that the precipitation of the cadmium begins with a molar relation of Cd: 2-sodium-aminobenzoate - 1 : 1,3 and is complete at a ratio of 1:3.5. A surplus of the reagent has no influence on the accuracy of results obtained. It was further shown, that it is innaterial for the accuracy of the determination how long the precipitate is left to set. The precipitate has well set after 10 to 15 minutes and is easily filterable. Repeated washing with water gives lower results and therefore the precipitate is washed with 70% alcohol, which does not cause any decrease of results even after 10 washings. The precipitation of cadrium according to this method occurs within a pH range of 2 to 6.5 quantitatively. The exactitude of determination is of the same order as that of the gravimetric determination of cadmium according to the sulfate method. With the aid of the method described the following salts of cadmium can to intermined: the acetate, the nitrate, the chloride and the issue of cadmium. Tests have shown that zinc is not precipitated by a solution of 2-sodiun-aminobenzoate. Therefore cadmium can be determined

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The Use of 2 Titrinetric 1	-Sodium-Paraaminobenzoic Acid for the 75-13-3-11/27 Determination of Cadmium
ASSOCIATION:	in the presence of zinc. It was shown, that, even in the pre- sence of a fivefold surplus of zinc satisfactory cadrium de- termination can be carried out satisfactorily with this new method. There are 8 tables and 10 references, 5 of which are Soviet.
	Farnatsevticheskiy institut, Odessa (Pharmaceutical Institute, Odessa)
SUBMITTED:	July 14, 1956
	1. CadmiumDetermination
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PORTNOV, A.I.; VASYUTINSKIY, A.I.

Use of substituted phenylarsonic and benzoic acids in inorganic analysis. Trudy kom. anal. khim. 11:192-197 '60. (HIRA 13:10)

1. Odesskiy farmatserticheskiy institut. (Benzenearsonic acid) (Benzoic acid) (Chemical tests and reagents)

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VASYUTINSKIY, A.I. [Vasiutyns'kyi, A.I.]; KHARLAMB, A.B., student

Refractometric analysis of anesthesine ointment. Farmatsev. zhur. (MIRA 14:5) 15 no.1:25-27 '60.

1. Kafedra farmatsevticheskoy khimii Zaporozhskogo farmatsevticheskogo instituta. (REFRACTOMETRY) (BENZOCAINE)

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Y CONTRACTOR CONTRACTOR OF A CONTRACTOR OF A CONTRACT OF A CONT - 1 AUTHORS: Amonenko, V.M., Vasyutinskiy, B.M., Lebedev, V.V. and TITLE: Vacuum Distillation of Metals with Condensation on a PERIODICAL: Fizika metallov i metallovedeniye, 1959, Vol 7, Nr 6, ABSTRACT: The properties of heat-resisting alloys are influenced to a considerable extent by the purity of the starting materials. Vacuum distillation is a promising way of purifying such materials. The authors describe their use for purifying iron of the method developed in 1952 at the Fiziko-tekhnicheskiy institut ANU&SSR (Physico-technical Institute, Ac.Sc. Ukrainian SSR) for vacuum distillation with condensation of the metal on a surface at a high temperature. The authors consider this more efficient than published methods and they have used it successfully for purifying beryllium (Ref 5). The distillation of the iron was effected in a working vessel (Fig 1) with evacuation by an oil diffusion pump (2500 litres/sec) and a type VN-2 backing pump. Card 1/4 0.5-3 litre alundum or beryllium-oxide crucibles wound

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SOV/126-7-6-9/24 Vacuum Distillation of Metals with Condensation on a Heated Surface

> with molybdenum or tungsten heating coils, contained the metal. The heated column directly over the crucible was generally lined with thin iron sheet, on which condensation occurred. The temperature of the column surface was chosen such that iron condensed while the impurities remained vaporized: the lower part up to 1300°C, the upper to about 1100°C. Assuming as a first approximation that the condensing metal and impurities form an ideal solid solution, the authors apply the Knudsen-Langmuir equation to calculate rates of evaporation. From a crucible at about 1580°C evaporation of metal occurred at 1 g/cm² hr., 75-80% of which was recovered at a column temperature of 1250-1300°C. Tables 1-3 show compositions before and after distillation (single and double) of armco, electrolytic (single only) and carbonyl irons, respectively. Purification from Mn, Mg, Cu, S, P, N $_2$ and O $_2$ was good and somewhat less so from aluminium. Considerable contamination from

Card 2/4 evaporation of crucible material was possible, but with double distillation the impurities could be reduced to

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sov/126-7-6-9/24 Vacuum Distillation of Metals with Condensation on a Heated Surface

The resistances of some long-needle single crystals of iron in the condensate were compared at 0°C and at low temperatures in the laboratory of B.G.Lazarev, acting member of the Ac.Sc. UkrSSR: the ratio values agree fairly closely (Table 4) with those of Meysner (Ref 6) for the purest iron and indicate that the needles were 99.996% Fe. The authors have also studied the purification of high-carbon (7% C, 73% Mn) and medium-carbon The same apparatus was used, evaporation temperatures being 1100-1400°C. Rates of evaporation tended to fall through impoverishment of surface layers with manganese and formation of a graphite layer. Lower iron contents were obtained when baffles (Fig 2) were fitted in the column. On the lower baffles, kept at about 1000°C, almost all iron condensed, the manganese condensing mainly on the middle baffles (750-800°C). Table 6 shows the composition of the condensate from the third and fourth baffles. A carbon content of under 5×10^{-3} % is The purity of the manganes; after a single inferred. Card 3/4 distillation is over 99.96%.

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SOV/126-7-6-9/24 Yacuum Distillation of Metals with Condensation on a Heated Surface There are 2 figures, 6 tables and 6 references, 3 of which are Soviet, 1 English and 1 French and 1 German. ASSOCIATION: Fiziko-tekhnicheskiy institut AN UkrSSR (Physico-Technical Institute, Ac.Sc. UkrSSR) SUBMITTED: July 22, 1957 Card 4/4

"APPROVED FOR RELEASE: 08/31/2001 CIA-RDP86-00513R001859020014-1 ENGERED AND AND A DEPARTMENT OF VASYIFTINSKTY, B.M.; KOGAN, V.S.; KARTMAZOV, G.N.; YAKIMENKO, L.F., Constitutional diagram of the nickel - chromium system. Fiz. met. i metalloved. 9 no. 4:558-563 Ap 160. (MIRA 14 (MIRA 14:5) 1. Fiziko-tekhnicheskiy institut AN USSR. (Phase rule and equilibrium) (Nickel-chromium alloys-Metallography)

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CONSISTERNO BARACON DESCRIPTION DESCRIPTION DESCRIPTION S TH. · VASYUTINSKIY 28441 s/185/61/006/002/016/020 D210/D204 17.4430 Amonenko, V.M., and Vasyutyns'kyy, B.M. AUTHORS: X The existence of a permeability minimum in the flow TITLE: of gases through porous films PERIODICAL: Ukrayins'kyy fizychnyy zhurnal, v. 6, no. 2, 1961, 263 - 266 TEXT: Beryllium films were obtained with the aid of precipitation of Be vapor on a molybdenum surface at 10^{-5} mm Hg. The temperature of the condensation surface was varied between 400 and 750°C in different experiments. Degassing of molybdenum before condensation affects essentially the character of distribution of the pores in the film. The permeability is defined by the authors as the ratio of flow through a plate to the flow through the same area in absence of the plate. Experimental determination of permeability was made according to the formula Uard 1/3

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 $\gamma = \frac{\ln \frac{\Delta P_1}{\Delta P_2} \cdot V_1 \cdot V_2}{s \sqrt{\frac{RT}{2\pi\mu}(V_1 + V_2)\Delta t}},$ (3)

 $\triangle P_1$ being the decrease of pressure through the film at the beginning of the experiment, $\triangle P_2$ the decrease at the end of the experiment, $\triangle t$ the time of variation of the decrease from $\triangle P_1$ to $\triangle P_2$, V_1 and V_2 the volumes separated by the film. Measurements were made on a device not essentially different from M. Knudsen's (Ref. 1: Ann. d. Physik, 28, 75, 1909), and graphs of the dependence of γ on P were traced. Only one graph is given by the authors (thickness of the film approx. 100 µ). From the relation $I_1 > 7 > r$ for the transition interval (7 - mean free path of the molecules, r - radius of capillaries, L not defined) it is estimated.

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The existence of a permeability ...

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ted that the mean radius of the capillaries is $r < 0.5 \mu$. From the existence of a minimum it is concluded that the pores are mostly direct canals. The permeability of films which had a minimum was always found to be small $(10^{-4} - 10^{-5})$. There are 2 figures and 6 references: 1 Soviet-bloc and 5 non-Soviet-bloc. The references to the English-language publications read as follows: W. Pollard, Phys. Rev., 69, 53, 1946; W. Pollard, W. Present, Phys. Rev., 73, 7, 762, 1948; Kormann, Proc. Roy. Soc., 203, 51, 1950.

ASSOCIATION: Fizyko-tekhnichnyy instytut AN URSR, ma Kharkıv (Institute of Technical Physics, AS UkrSSR, Kharkov)

SUBMITTED: July 8, 1960

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22966 5/126/61/011/005/014/015 18.8200 E193/E183 Gindin, I.A., Starodubov, Ya.D., and Vasyutinskiy, B.M. Plasticity and brittleness of cast melybdenum at AUTHORS : temperatures between 4.2 and 700 $^{\circ}K$. PERIODICAL: Fizika metallov i metallovedeniye, Vol.11, No.5, 1961, TITLE: The object of the present investigation was to explore pp. 794-800 the possibilities of low-temperature application of refractory metals such as Mo, Cr, W, Nb, etc. To this end, the mechanical properties of Mo were datermined by means of the standard tensile test at 4.2-700 $^{\circ}K_{1}$ and the effect of preliminary heat- and mechanical treatment on the transition temperature from the ductile to brittle fracture was studied. Mo of 99.95% purity was used in the experiments, the main impurities consisting of (%): 0.005 Fe; 0.01 N1; 0.017 Ca; 0.002 A1; 0.002 0; 0.0009 N; 0.0006 H. To ensure uniform grain size, the ingots cast in vacuum-arc furnace were hot-rolled at 1000 °C to 50% reduction in thickness, sparkmachining having been used for the preparation of flat, tensile test pieces of 7 mm gauge length and 2 mm² cross-section. Card 1/6 and the production of the president residence in the second second second second second second second second s

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22966 s/126/61/011/005/014/015 Plasticity and brittleness of cast E193/E183 After machining the test pieces were vacuum-annealed at 1280 $^{
m o}$ C This treatment reduced the gaseous impurity content and produced a coarsely-crystalline structure with the average grain size of 200-400 μ . The tensile tests were carried out at 4.2, 20, 77, 183. 200, 223, 243, 300, 435 and 700 °K; at two rates of strain, 0.4 and 30 μ /sec. Some of the results obtained at the rate of strain of 0.4 μ /sec are reproduced in Fig.3, where the yield point (σ_g) , U.T.S. (σ_b) and the true tensile strength (σ_u) measured in kg/mm² are plotted against the test temperature (°K). It will be seen that all these properties increase with decreasing temperature. The point of intersection of the σ_B and σ_b curves determined the transition temperature from ductile to brittle fracture, which in this case was 183 °K. The unusual feature of curves shown in Fig.3 is that they all pass through a maximum at approximately 80 °K, since it is generally believed that the tensile strength in the brittle fractural region does not depend on temperature. increasing rate of strain, both σ_s and σ_b increased, and the With temperature of the transition from ductile to brittle fracture was shifted to 208 °K. The plastic properties of Mo have been found to decrease with decreasing temperature at a rate which increases with Card 2/ 6 -CONTRACTOR OF THE OWNER AND THE OWNER OF THE OWNER

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22966 s/126/61/011/005/014/015 Plasticity and brittleness of cast E193/E183 increasing rate of strain. This is illustrated in Fig.5, where elongation (5, %) and reduction of area (ψ , %) are plotted against the test temperature (°K) for specimens extended at 0.4 (open circles and squares) and 30 μ /sec (black circles and triangles). In the second stage of the present investigation, the tensile test pieces were subjected to the following treatment: (1) loading at room temperature and at a rate of strain of 0.4 μ /sec to attain a stress equal to 0.5 σ_{B} ; (2) slow cooling under constant load to 77.2 °K and holding at that temperature for 1-1.5 It was found that after this preliminary treatment, the test hours. pieces tested at 183 °K (i.e. at the critical temperature) exhibited some degree of ductility (\$ 5%). Fig.6 shows the actual load (kg) versus strain (μ) curves for Mo tested at 183 °K at a rate of strain of 0.4 μ /sec for untreated (curve 1) and treated (curve 2) specimens. In Fig.7 the elongation (δ , %) of untreated (curve 1) and treated (curve 2) test pieces is plotted against the test temperature. It was found also that no significant improvement in ductility can be achieved by cooling the metal (during the treatment described above) to temperatures lower than 77 °K. An increase in Card 3/6 STREET STREETS 0712-07-07-07

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AUTHORS:	2808, 2208, 1418, 1416 25923 Gindin, I.A., Staradubov	, Ya.D., Vasyutinskiy, B.M.
TITLE:	Metallographic investiga tension at 4.2 to 700°K.	tion of molybdenum deformed in II
PERIODICAL:	Fizika metallov i metall pp.132-139	ovedeniye, 1961, Vol.12, No.1,
changes occ but also in temperature the results undertaken molybdenum electron mi	ur not only in the mechan its microstructure. Sis s lower than 77°C had bee of which are described is with the object of studyis deformed in tension at 4, croscopes were used in the was used, the changes in	of this transition is approached nical properties of the metal ince no study of molybdenum at en reported, the investigation, in the present paper, was ing the microstructure of ,2 to 700°K. Both optical and the examination of the specimens. the microstructure on the face having been revealed with

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in different grains, mean magnitude of absolute displacement in slip, and the dependence of these characteristics on the temperature and degree of plastic deformation was evaluated. The (1) At all temperatures at results can be summarized as follows. which molybdenum remains plastic (that is down to $183 \, {}^{\circ}\overline{K}$) it deforms plastically by the mechanism of slip. As in other body-centred cubic metals, branched slip lines are formed on molybdenum, indicating a more complex mechanism of deformation than that obtaining in face-centred cubic metals. This shape of the slip lines can be observed already in the early stages of plastic deformation corresponding to an elongation of $\delta = 1 - 2\%$. The effect becomes more pronounced with increasing degree of deformation at any given temperature but the effect of heavy deformation is most pronounced near the ductile-to-brittle transition temperature. Fig.2 shows (magnified 330-fold) the microstructure (a) and the interference pattern (b) of the slip bands formed on molybdenum deformed at 200°K to 8 = 0.8%; the magnitude of the absolute slip was in this case approx 0.25 μ_{\star} In suitably oriented grains (particularly at high temperatures) a Card 2/9

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system of intersecting slip lines is formed. Increasing the degree of deformation of molybdenum at 240 to 700°K brings about the appearance of new slip bands and an increase in the displacement along the slip planes. The development of the process of deformation, however, is manifested predominantly by growth of the Thus, for example, just before the initially-formed slip bands. fracture of a specimen ($\delta = 38\%$) at 700°K, the slip bands may become 6 to 7 µ wide. The density of the slip lines also changes At 700°K, it is relatively small and slip bands, with temperature. At 300°K, the density of slip spaced at 12 to 15 μ , predominate. bands corresponding to the same degrees of deformation is higher, the width of the slip bands and the spacing between them decreasing. With a further decrease in temperature, the density of slip bands again decreases approaching that obtaining at 700 °K. (2) In addition to deformation by slip (as revealed by the formation of slip bands) plastic deformation of molybdenum at room temperature entails a specific mode of deformation, localized at the grain boundaries and in the grain-boundary regions. This mechanism operates at relatively low strains (3 to 5%). With increasing strain some of the regions of localized deformation grow Card 3/9

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in size and cracks are formed at the boundaries of these regions after heavy deformation. The width of these near-boundary regions can reach 25 to 30 μ , the relative displacement of adjacent grains along the grain-boundary being several tenths of a μ_{*} This mode of plastic deformation which has been observed in pure iron at sufficiently low temperatures (Ref.4: Gindin I.A. and Starodubov Ya.D. FTT, 1959, 1, 1794) appears to be a property of The microstructure and interference pattern of the pure metals. grain-boundary and the grain-boundary region of molybdenum, deformed at 300 K to 5 = 20%, is shown in Fig.5a and 5b respectively (3) With decreasing temperature the (magnified 440-fold). character of plastic deformation changes considerably. At temperatures approaching the ductile-to-brittle transition, fragmentation and block formation precede the appearance of slip bands. The formation of blocks (whose size, determined with the aid of an electron microscope, was found to be $(2-3) \times 10^{-4}$ cm) increases the resistance of molybdenum to slip and twinning; the process of deformation becomes less uniform and fracture takes place at relatively small strains. (4) In contrast to other metals with Card 4/9

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body-centred cubic crystal structure, twinning plays a relatively insignificant part in the plastic deformation of molybdenum. Thin twins (1 to 2 μ thick) appear in specimens deformed below 246 K but only in isolated grains. An electron microphotograph (magnified 11250 times) of a twin (approx 0.5 μ thick) in molybdenum deformed at 200°K to & 2% is shown in Fig.8. A specific characteristic of twins of this type is the presence of lightly and heavily distorted zones showing, respectively, as dark and light bands on the microphotograph. It is postulated that the highly distorted zone is formed suddenly when a certain stress, required to initiate the process of twinning, is reached. The appearance of this zone is accompanied by the formation of a mosaic structure in the boundary region and by the formation of blocks and their elastic recovery. As in the case of iron, growth of a twin in molybdenum takes place by movement of one of its boundaries; on reaching the distorted region, the growth of the twin ceases owing to the strain-hardening of this zone. (5) The specific character of plastic deformation of molybdenum is reflected in the manner in which this metal fractures. At 300 and 700°K fracture takes place along the slip planes and a well-defined neck is formed Card 5/9

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in a tensile test piece. Cracks along the slip planes appear also in molybdenum, tested at 240°K, but in this case they are accompanied by cracks along the cleavage planes, the number of This is these cracks increasing with decreasing temperature. illustrated in Fig.9 (magnified 440-fold) showing a portion of a test piece deformed at 243 K to $\delta = 18\%$ in which the parallel slip lines end at a crack along the cleavage plane. On approaching the ductile-to-brittle transition temperature, and particularly below Side it, cracks along the grain- and block-boundaries are formed. by side with the main crack a number of cracks parallel to it but not traversing the entire cross-section of the test piece can be observed. Fracture below the critical temperature is both transand inter-crystalline, although the latter is relatively less The decrease in strength of molybdenum below 27°K pronounced. has been attributed to the formation of a large number of surface The formation of the cracks which cause premature fracture. surface cracks is, in turn, associated with a high concentration of oxygen in the surface layer. It was concluded from the results of the present investigation that the character of plastic deformation of 99.95% molybdenum in the temperature interval Card 6/9

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Vesyutinskiy B.M., Kartmazov, G.N., Finkel' V.A. The structure of chromium in the temperature range of AUTHORS : TITLE : PERIODICAL: Fizika metallov i metallovedeniye, v.12. no.5 1961. 700 - 1700[°]C Previous investigations of the crystalline structure of Ni-Cr alloys in the temperature range of 1840 to 1930°C indicated the presence of a cubic, face-centred β -phase of chromium and the possibility was suggested that chromium exists in five allotropic modifications. To verify these assumptions a study was made of the crystalline structure of 80 x 10 x 2 mm The examination was chromium specimens with a purity of 99.94%. made in a high-temperature X-ray camera, which was first evacuated to 10^{-4} mm Hg and then filled in with argon to the pressure of The specimen was heated by passing through it electric current (up to 600 A). The examination was made in

Cr anticathods radiation at a specimen-film distance of 60 mm. which ensured good resolution of the a-doublet and enabled an The experimental accurate determination of the lattice parameter. Card 1/42 3

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set-up was designed to facilitate reflection from the (211) plane of the body-centred chromium in the whole interval of the test temperatures. The experimental results are shown in Fig.1 and 2. The results shown in Fig.1 indicate the absence of any polymorphi: transformations in pure chromium within the temperature range: investigated. There are 2 figures and 14 references: 11 Soviet-bloc and 3 non-Soviet-bloc. The three references to English language publications read as follows: Ref.1: Bloom D.S., Grant N.J. J. Metals, v.3 (11), 1951, 1009 Ref.2: Abrahamson E., Grant N.J. J. Metals, v.8, 1956, 975. Ref.3: Stein C., Grant N.J. J. Metals, v.7, 1955, 127.

ASSOCIATION Fiziko tekhnicheskiy institut AN UkrSSR (Physicotechnical Institute AS UkrSSR)

SUBMITTED: April 19, 1961

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5778 s/180/62/000/001/014/014 E040/E135 Azhazha, V.M., Vasyutinskiy, B.M., and Shapoval, B.I. 18.1410 AUTHORS : Mechanical properties of high purity nickel (Khar'kov) PERIODICAL: Akademiya nauk SSSR. Izvestiya. Otdeleniye TITLE: tekhnicheskikh nauk. Metallurgiya i toplivo. no.1, 1962, 160-161 Nickel of 99.98% purity (0.006% Fe, 0.003% Si, 0.001% Al) was produced using a vacuum distillation column described previously by V.M. Amonenko et al. (Ref.1: Fizika metallov i metallovedeniye, v.7, no.6, 1959, 369). Spectrochemical analysis data showed that the nickel prepared by vacuum distillation from alundum crucibles at 1480 °C had the impurities of Bi, Pb, S, Mg, Sb, Cd and P not exceeding 3×10^{-3} %. Mechanical strength tests were carried out on specimens prepared from distilled nickel subjected subsequently to re-melting under vacuum. The test castings were then coldworked to 40-50% and annealed in order to produce a grain diameter of 0.06-0.08 mm. Tensile strength specimens were X Card 1/3

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Mechanical properties of high ...

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30 mm in length and 6 mm in diameter. The specimens were annealed after polishing in order to remove the effects of coldworking. It was found that in tensile tests at 20 °C highpurity Ni behaves in the same manner as pure polycrystalline Al, i.e. after formation of one or several necks further deformation proceeds by slip. Impact strength was > 36 and > 34 kgni/cm² at 20 and -196 °C, respectively, compared with 18.9 kgm/cm² for commercial grade Ni. Specimens were not fractured but on those tested at -196 °C clearly visible cracks were observed. The room-temperature tensile strength was 34.0 kg/mm²; yield strength 6.7 kg/mm²; elongation 63%; and Brinell hardness 56 to 58 kg/mm². Because magnetic properties of ferromagnetic materials depend on the presence of impurities, especially gases, the ferromagnetic anomaly of ductility of high-purity nickel, which contains only a negligible quantity of gases, was expected to be indicated more clearly than in ordinary purity nickel. There are 3 figures and 2 tables.

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Nechanical properties of high ... S/180/62/000C/C01/014/014
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ASSOCIATION: Fiziko-tekhnicheskiy institut AN USSR
g. Khar'kov
(Physico-technical Institute AS Ukr.SSR, Khar'kov)
SUBMITTED: September 14, 1960
Card 3/3

CIA-RDP86-00513R001859020014-1

artist store to a - · ÁH S/185/62/007/006/010/014 D407/D301 Vasyutyns'kyy, B. M., Kartmazov, H. M. and Finkel', AUTHORS: V.O. X-ray investigations of the structure of tantalum up TITLE: to 2700°C Ukrayins'kyy fizychnyy zhurnal, v. 7, no. 6, 1962, PERIODICAL: 661-662 TEXT: In the present investigation, the temperature range is extended beyond 2200°C. The specimens were made of tantalum wool, 0.3 mm thick. The metal was annealed in a vacuum at 2200°C so as to remove gaseous impurities. The X-ray analysis was carried out in a high-temperature X-ray chamber in a vacuum of $1-3\cdot10^{-4}$ mm Hg. The specimens were heated by an electric current. Only the tantalum line was observed over the entire temperature range; the parameters of the body-centered cubic lattice vary smoothly with temperature. This indicates the absence of phase transitions in tantalum. The temperature dependence of the lattice parameters of tantalum Card 1/2sour lasa

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CIA-RDP86-00513R001859020014-1 "APPROVED FOR RELEASE: 08/31/2001 THERE PERMIT METERS AND AND A DESCRIPTION OF A DESCRIPTIO OVERTER SOUTHINGSED STREET S/185/62/007/006/010/014 X-ray investigations ... D407/D301 can be expressed by the empirical formula $a_{mOC} = (3.3017 + 1.4142 \cdot 10^{-5} T + 0.8660 \cdot 10^{-8} T^2) kX.$ The coefficient of linear expansion was calculated by an approximate formula. The temperature dependence of the coefficient of linear expansion is shown in a figure. At temperatures above 2000°C the lattice parameter varies almost linearly with temperature, whereas the coefficient of linear expansion remains practically unchanged. There are 2 figures. The most important English-language reference reads as follows: J. W. Edwards, R. Speiser, H. L. Johnson, J. Appl. Phys., 22, 424, 1951. ASSOCIATION: Fizyko-tekhnichnyy instytut AN UkrRSR, Kharkiv (Physicotechnical Institute of the AS UkrRSR, Kharkiv) February 1, 1962 SUBMITTED: Card 2/2

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s/126/62/013/002/017/019 E039/E135 18.1731 Vasyutinskiy, B.M., Kogan, V.S., Kartmazov, G.N., AUTHORS : and Yakimenko, L.F. The formation of textured layers of nitride on chromium obtained by condensation in vacuum from TITLE: the vapour phase PERIODICAL: Fizika metallov i metallovedeniye, v.13, no.2, 1962, 310-311 It is shown that the skin formed on the surface of chromium when heated in air or oxygen consists of two layers: an external layer of rhombic Cr_203 and an internal layer of hexagonal Cr₂N. This was discovered by means of X-ray diffraction measurements. The structure of the skin formed on chromium when heated in air and in nitrogen up to 1300 °C was examined for two different samples; one was chromium cast and rolled in vacuum, and the other a sample of chromium obtained by condensation from the vapour phase. This condensation was carried out at a pressure of 10-3 mm Hg on to a molybdenum plate over a period of Card 1/2 后期制度为法 2364264

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10-15 hours forming a layer 300-500 μ thick. This layer was then annealed in air for 450 hours. The skin formed was studied by means of X-ray diffraction using K_{α} -Cr radiation to improve definition. Maximum reflection from the (110) plane was obtained with the sample placed at 56° to the incident beam, indicating that the nitride is orientated with the (110) plane parallel to the surface. In the case of chromium cast and rolled in a vacuum at a temperature of 1100 °C no structure corresponding to the nitride layer was discovered; similarly, chromium cast and rolled in air and in nitrogen at a temperature of 900-1200 °C showed no structure. It is observed that the structured layer of nitride on the chromium condensed from the vapour phase is much more firmly bonded to the outer oxide layer than in the case of the structureless nitride on cast chromium from which the oxide layer is easily separated. ASSOCIATION: Fiziko-tekhnicheskiy institut AN UkrSSR

(Physicotechnical Institute, AS UKrSSR) SUBMITTED: May 22, 1961

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VASYUTINSKIY, B.M.; KARTMAZOV, G.N.; FINKEL', V.A.

Obtaining filiform crystals of chromium. Fiz.met.i metalloved. 14 no.5:792-793 N '62. (MIRA 15:12)

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1. Fiziko-tekhnicheskiy institut AN UkrSSR. (Chromium) (Crystallization)

4. 日本**的一种主义**的保护的资源并且在全国的主义。

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VASYUTINSKIY, B.M.; KARTMAZOV, G.N.

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Mechanism of nickel oxidation. Fiz.met.i metalloved. 15 no.1: 132-134 Ja 163. (MIRA 16:2)

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AMONENKO, V.M., <u>VASYUTINSKIY</u> , B.M.; FINKELI, V.A.	KARTMAZOV, G.N.; SMIRNOV, Yu.N.;
Investigating the structure Fiz.met.i metalloved. 15 no.	of tantalum at high temperatures. 3:444-450 Mr '63. (MIRA 16:4)
 Fiziko-tekhnicheskiy inst (Tantalum-Testing) 	itut AN UkrSSR. (Metals at high temperatures)
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CIA-RDP86-00513R001859020014-1 "APPROVED FOR RELEASE: 08/31/2001 CONTRACTOR DE LA CONTRACTORIZIÓN DE LA CONTRACTOR DE LA C 的目的形式的问题。但是我们的问题,我们是我们的问题,我们是我们的问题,我们就是我们的问题。 EWT(m)/EWP(w)/EWA(d)/EWP(t)/EWP(b) IJP(c)/ASD(m)-7 50/50 L 17709-65 S/0126/64/017/006/0892/0897 ACCESSION NR: AP4042048 AUTHOR: Vasyutinskiy, B., H.; Kartmazov, G. N.; Papirov, I. I. TITLE: Surface structure and properties of acicular crystals of Ē chromium A SOURCE: Fizika metallov i metallovedeniye, v. 17, no. 6, 1964, 892-897 TOPIC TAGS: acicular crystal, whisker surface structure, chromium strength, growth mechanism, chromium 18 Pointing out that the formation and growth of acicular ABSTRACTI crystals is far more complex than indicated by earlier investigators. the suthors discuss surface structure, strength characteristics, and growth mechanism of Cr crystals with a maximum cross section of 20×10^{-6} mm² produced by vacuum condensation. Examination under a standard optical microscope showed that crystals grow in steps; the mean height of individual steps varied from 1700 to 2300 A. The growth begins at the basal face and proceeds toward the apex. Under 1/3 Card

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an electron microscope stepped and complex profiles as well as smooth surface regions were also identified. Failure in tensile tests occurred by a cleavage and was not preceded by conspicuous plastic deformation, Investigations showed that the thicker crystals had a lower strength. The strength reached 590 kg/mm² in crystals with macroscopic growth steps. Since appreciable strength was observed in crystals both with a smooth surface and with growth steps, the authors assume that the latter are not formed as a result of dislocation. The discontinuation of axial growth is attributed to the mechanism of the development of steps on the lateral face. Apparently, with each new layer that forms at the basal face and reaches the apex, the supply of crystallizing substance to the mobile whisker end is reduced by the diffusion of particles toward the step of the next layer. This was corroborated by the fact that thin crystals possess a smaller number of growth steps than thick specimens formed under analogous conditions. Moreover, the same crystal may be rather smooth near the apex and rough in the thicker portion. The changes in the axial growth under conditions of temperature control were caused by the changes in the growth mechanism. The contribution of A. A. Chernov is gratefully acknowledged. Orig. art. has: 4 figures, 2 formulas, and 1 table. 2/3 Card

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and the second L 17700-65 ACCESSION NR: AP4042048 ASSOCIATION: Fiziko-tekhnicheskiy institut AN UkrSSR (Physicotechnical Institute, AN UkrSSR) SUB CODE: SS, IC ENCL: 00 SUBMITTED: 03Ju163 OTHER: 010 NO REF SOV: 005 Card 3/3

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CCESSION NR. AP5017281		'\$\$\0181765700770077194471	1951
JTHOR: Shvachko, V. L.; Na artmazev, G. N. ITLE: Using secondary <u>ion-</u>	1.		'
ith the surface of <u>niobium</u>	- 27		27
NURCE: Fizika tverdogo tel	•]	944-1951	
OPIC TAGS: ion emission, n	iobium, oxidation		4
BSTRACT: The method of sec omposition of <u>oxides</u> which xygen. It was established owing oxides form on the cu ressure of approximately 10 emperature interval from 12 b ₂ O ₃ . The corrosion wear of ion of NbO starts at 140000 he 20-800000 temperature ran ion NbO ₂ • NbO+0 with the	form on the surface of that in the temperature rface of nioblum which "4 mm of mercury: NBO, 00 to 2000°C the surfac f Nh results from the f and then increases ver ge NBO; indergnes decom	nioblum when it interacts range from 20 to 120000 f is in an oxygen atmosphere NBO2, NB2O3, and NB2O5. We of NB contains only NBO ormation and subsequent ev v rapidly with temperature position according to the	with the fol- e at a In the and vapora- e. In reac-
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AUTHOR: Ivanov, V. Ys.; Nec. A. D.; Vasyutinskiy, B. M.;	hiporenko, Ye, P. (Dr. of Technical Sci Kartmazov, G. N.	ences); <u>Osipov</u> , 71 671
ORG: none		D7 [
TITLE: Thermal stresses in	chromium coatings on molybdenum 27	
SOURCE: <u>Seminar po sharosto</u> pokrytiya (Heat-resistant co 1965, 77-82	ykim pokrytiyam. Leningrad, 1964. Zharo patings); trudy seminara. Leningrad, Izd	stoykiye I-vo Nauka
stress, adhesion, thermal st		
by the condensation in vacuu protective properties, but t	(FMM, IX, 4, 558, 1960) showed that comum of Cr <u>vapors</u> on the surface of Mo surthat their service life decreased considerature fluctuations. A study was made stresses in chromium coatings on molybe	Jerably when on the effect
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in various vacuum conditions and having various strengths of adhesion of the coating to the substrate. The value of stress (σ) was determined from changes in the deflection (d) of the plated samples (loo x 5 x 2mm) by using the formula $\sigma = 4 \text{ E h}^2_2 d/3l^2h_1$ ($h_1 + h_2$), where E is the Young modulus, l is the length of the coated part of the sample, and h_1 and h_2 are the thicknesses of the coating and the base metal, respectively. The curves of deflection (in rm) vs temperature were plotted during the experiments. The changes in the slope of the curves (inflections), corresponding to the conversion of elastic into nonelastic deformations, were observed during heating and cooling of the samples. Nonelastic deformations in the low-temperature range (≤ 4000) were formed at the critical stress $\sigma = 8 \text{ kg} / \text{ mm}^2$. The value of the curves with the coatings of various degrees of the curves deformation strength of adhesion was changed by applying the coating to the substrate. Experiments with the coating the roughness to the surface of Mo oxidized to various degrees, or by increasing the roughness of the No surface) proved that the value of the critical stress did not depend of the Mo surface of strength of coatings applied to the surfaces of oxidized to various degrees or by increasing the roughness of the No surface proved that the value of the critical stress did not depend of the substrate. The adhesion strength of coatings applied to the surfaces of oxidized to various degrees.

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	a) 0 S/0286/63/0 B. M.; Rogan, V. S.; Laz	00/010/0072/0072
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Lazareva, L. S.	and an and a second	. 15
	graphite. Class 48, No.	154752
TITLE: TINPIALING 91 4		10, 1963, 72
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1		ing, vacuum tinning
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ABSTRACT: A patent ha	s been issued for a metric them in molten tin. To tinning process is carrie	obtain a high-
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1000C with a maximum o	them in molten tin. To tinning process is carrie f 0.01% tungsten molybde rbide-forming metals add	ed to the tin bath.
zirconium, or other ca	rbide-torming metals act	
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ASSOCIATION: none SUBMITTED: 21Jun62 SUB_CODE: ML	DATE ACQ: 140ct63 No Ref Sov: 000	OTHER: 000

	I. $101,411-67$ EWT(m)/EWP(t)/ETI IJP(c) JD/JQ ACC NR: AP6023705 SOURCE CODE: UR/0126/66/021/004/0620/0621 53
Ū.	AITTHORS: Vagyutinicity, B. M.; Kartmazov, G. N.; Smirnov, Yu. N.; Finkel', V. A.
	ORG: <u>Physico-Technical Institute</u> , <u>AN UkrSSR</u> (F'ziko-tekhnicheskiy institut AN UkrSSR) TITLE: Investigation of the crystalline structure of niobium and vanadium at high temperatures (6 27
	SOURCE: Fizika metallov i metallovedeniye, v. 21, no. 4, 1966, 620-621
	TOPIC TAGS: niobium, vanadium, x ray spectroscopy, crystal lattice parameter
	ABSTRACT: The crystal structure of niobium and vanadium was determined as a function of the temporature. The experimental procedure was described earlier by V. M. Amonenko, B. M. Vasyutinskiy, G. N. Kartmazov, Yu. N. Smirnov, and V. A. Finkel (FMM, 1963, 15, 444). The experimental results are presented graphically (see Fig. 1). It was found that the temporature dependence of the lattice parameters obeyed the
	following relationship
	$a_{T^*0}^{N_0} = 3,3001 (1 + 7,223 \cdot 10^{-0} \cdot T + 7,667 \cdot 10^{-10} \cdot T^*) \hat{\Lambda};$ $a_{T^*0}^{Y} = 3,0206 (1 + 7,314 \cdot 10^{-0} \cdot T + 2,044 \cdot 10^{-10} \cdot T^*) \hat{\Lambda}.$
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ACC NR: AT7004209 AUTHORS: Kogan, V. S.; Vasyutinskiy, B. M.; Lazarev, B. G. ORG: none	27
TITLE: Studying phase diagrams with the use of diffusion layers, SOURCE: AN SSSR. Institut metallurgii. Eksperimental'naya tekhnika i metody vysokotemperaturnykh izmereniy (Experimental techniques and methods of high temper- ature measurement). Moscow, Izd-vo Nauka, 1966, 121-127	
TOPIC TAGS: metal phase system, metal vapor deposition, metallographic examination, nickel, chromium, molybdenum, niobium, tin, iron, tantalum ABSTRACT: The obtaining of metal phase diagrams by a multilayer technique is described. The technique, an extension of the work of L. S. Palatnik, V. M. Kosevic and L. V. Tyrina (FMM, 1961, 11, 229), consists of condensing an appropriate metallit condensate. This technique was applied to the study of the phase diagrams of the following systems: Cr-Ni, Nb-Sn, Fe-Ta, and Mo-Cr-Ni. The experimental results, shown graphically (see Fig. 1), were published earlier in three communications by pictures were taken and the microhardness of the specimens was determined. The results are shown graphically. It is concluded that the condensation-diffusion layer Cord 1/2	h, c

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NEW THE PROPERTY SERVICE STREET, S TEAL STREET, S VASVUTINSKIY, N.A. (Kerch'); GLE MLV, S.K. (Kerch') Reduction of Kerch iron ores containing organic substances. Izv. Al S.S. Ctd. teki. nauk. Met. i topl. no.1:149-151 Ja-F (MIFA 14:2) '61. (Kerch Peninsula--Iron ores) (Iron-Motallurgy) THE R. LOW CO.

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Optimum degree of reduction in the magnetizing roasting of (MIRA 14:10) iron ores. Stal' 21 no.10:878-881 0 '61.

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1. Kamyshburunskiy zhelezorudnyy kombinat. (Cre dressing) (Iron ores)

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Magnetite from agglomerates of Kerch iron ores. Zap.Vses.min.ob-va 90 no.6:731-735 °61. (MIRA 15:2) (Kerch Peninsula--Magnetite)

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1.1.1.2 T MANUTEBELY, H.A.; HEVOXSA, G.C. Some specific features of armonic distribution in the iron cross of Kerch. Do'1, AN SSEN 341 No.1:107-00 N 163. (TTE: 14:11) 1. Desychoromaki; shelerereir; orbirat. Fredstavlene a'u.douileo. I.N. Strait.ovyn. (Merch Foninsule--Iron eres) (Arconic)



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GREBNEV, S.K.; VASYUTINSKIY, N.A.

Specific magnetic susceptibility of powders with a various magnetite content. Obog.rud. 7 no.1:54-55 '62. (MIRA 15:3)

1. Kerchenskaya obogatitel'naya fabrika. (Metal powders--Magnetic properties) (Magnetite)


VASYUTINCKLY, N.A. (Zeporozh'ye)

Characteristics of the reduction of Kerch ores. Izv. AN CSSR. Met. i gor. delo1:20-25 Ja-F 164. (MIRA 17:4)

VASYUTINSKIY, N.A. (Zaporozh'ya)

PERSONAL PROPERTY AND ADDRESS AND ADDRESS

Conditions for the formation and the thermal stability of maghemite. Izv. AN SSSR. Otd. tekh. nauk. Mat. i gor. delo ho.4:39-45 Jl-Ag (MIRA 16:10) '63.

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L 3498 66 EWT(m)/T/EWP(t)/EWP(b)/EWA(o) ACCESSION NR: AP5024860	UR/0136/65/000/010/0070/0071
AUTHOR: <u>Vasyutinskiy, N. A.</u> ; Rys'yeva, Y P.; Kushkin, B. N. Y4,55 TITLE: Metallographic investigation of p sponge	v. I.; Rodyakin, V. V.; Chernysheva, S. 44.5 vorosity in magnesium-reduced <u>titanium</u> 44.5 7 27
SOURCE: Tsvetnyye metally, no. 10, 1965 TOPIC TAGS: titanium, sponge metal, por	ous metal, porosity, metal grain struc
ABSTRACT: The structure of the titanium sium affects markedly the process of the ticularly the degree of elimination of c	-
Card 1/3	

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L 3498-66 ACCESSION NR: AP5024860 vestigation by means of which new findings on this subject have been obtained. Specimens of Ti sponge were microscopically examined following their treatment with pore-filling rosin and subsequent polishing with abrasive powders and etching for 1 min in a solution of 10 cc HF, 30 cc HNO_3 and 50 cc H_2 at room temperature for 1 min. The specimens pertained to three different sponges produced at different rates of feed of TiCl₄ to the reactor. Findings: in sponge 1 (TiCl₄ feed rate: 150 kg/m²-hr) irregularly shaped pores of from 40-60 to 100-150 µ predominate, with most of the pores having smooth (round) contours: in sponge 2 (TiCl₄ feed rate: 230 kg/m²₂-hr) the micropore size is more uniform; in sponge 3 (TiCl₄ feed rate: 320 kg/m²-hr) the micropore size is from 5 to 250 μ and the size distribution is as irregular as in sponge 1. On the whole, sponge porosity increases with increasing TiCl4 feed rate, while at the same time the character of pores changes -- they become more irregularly shaped, with "lacerated" contours. This indicates an increase in the crystallization rate of Ti and a decrease in the effectiveness of recrystallization processes. Sections of sponge 1 reveal two basic structural varieties of the α -modification of Ti -- polyhedral (mostly equiaxial from 20-30 to 100-150 μ) and elongated acicular grains; this pattern is less distinctive for sponge 2. The visually observable dendrites of the titanium sponge proved, on microscopic examination, to have a polycrystalline structure, they 2/3 Card

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clearly underwant complete recr dendritic structures have survi rate of feed of TiCl ₄ to the re of porosity of the sponge but a structure of Ti itself. It show subjected to the vacuum separat	actor not only alters the lso is accompanied by char id be considered that the	extent and character ages in the micro- sponge investigated was changes in sponge stru
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subjected to the vacuum separate ture that were caused by change extent by the subsequent change ration. Orig. art. has: 2 figur ASSOCIATION: none SUEMITTED: 00	ENCL: 00	on were offset to some sponge during its sepa-





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VASYUTINSKIY, S. B.

The following is among dissertations of the Leningrad Folytechnic Institute imeni Kaknin:

NAMES OF THE OWNER OF

"Electromagnetic Processes of Rectifying and Inverting Circuits of Ionic Inverters with the Use of Filters in the Circuit of the Rectified Current." 12 December 1949. An examination is made of current rectifying and inverting processes in an invertor is connected, in accordance with a threephase bridge circuit with bunching parameters. In the circuit of the pulsating current and which is provided with parallel-connected, two-element resonance circuits on the side of the pulsating current.

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VASIUTINSKIN, S.B.

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HARRING THE STOL PROPERTY.

AUTHOR: Vasyutinskiy, S.B., Candidate of Technical Sciences and Pagayenko, G.P., Engineer.

The Thermal Design of Current-carrying Pipes. (Teplovoy raschet tokovedushchikh trubok) TITLE:

PERIODICAL: Vestnik Elektropromyshlennosti, 1957, Vol.28, No.10,

ABSTRACT; Hollow pipes are often used as current conductors where they can reduce the skin effect or are required for the passage of water or other fluids for cooling. At the present time there is no complete thermal design procedure for currentcarrying pipes with liquid cooling that takes account of such factors as heating of the liquid along the pipe, the change of the resistance of the material of the pipe with temperature along its length and the thermal conductivity of the pipe. The procedure here proposed is valid with any cooling fluid and in particular water. The steady state conditions of a hollow pipe of given external and internal diameters and length are considered. It is assumed that all the heat generated in the pipe is removed by the water and that there is no heat exchange between the pipe and the surrounding medium. An equation is formulated for the quantity of heat gener-Card 1/3 ated in unit time in a short element of pipe and a heat

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The Thermal Design of Current-carrying Pipes. 110-10-8/18 balance equation is derived. The equations are solved and it is shown that thermal design of current-carrying pipes with allowance for thermal conductivity is not fundamentally difficult although it requires laborious calculations. The presence of thermal conductivity leads to some equalisation of temperature along the length of the pipe, which is small when the pipe is long and of small cross-section. With pipes that are several metres long and with a cross-section of several square mm the influence of thermal conductivity may be neglected. This simplifies the equations and it is shown that in this case the temperature distributions in the pipes and water follow an exponential law. The expressions can be still further simplified if the temperature coefficient of resistance of the pipe is ignored and the specific resistance is chosen to correspond with the temperature of the middle of the pipe. The procedure described was used to make a large number of thermal calculations on current-carrying pipes. It may be concluded that, by neglecting the change of specific resistance with temperature along the pipe, errors of not more than 10% are introduced if the specific resistance is melated to the card 2/3 arithmetic mean temperature of the pipe. If the thermal

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conduc nct mor making pipes tions ture of and in made f dix g: kind of	Design of Current-carrying Fipes. ctivity of the material is ignored th re than 1 - 2%. It is therefore reco g preliminary thermal calculations or use be made of the existing procedur made for checking purposes should al change of specific resistance of the n particularly important cases allows for the thermal conductivity of the m ives a numerical example of a thermal described. is 1 figure.	mmended that in current-carrying e, whilst calcula- low for the tempera- material of the pipe ince should also be material. An appen-
ASSOCIATION	: Leningrad Polytechnical Institute Politekhnicheskiy Institut)	(Leningradskiy
SUBMITTED:	September 11, 1956	
AVAILABIE: Card 3/3	Library of Congress	

CIA-RDP86-00513R001859020014-1 - A PRIME TO PERSON REPRESENTATION REPRESENTATION

VASYUTINSKIY, S.B.; NAGAYENKO, G.P.

Designs and basic equations of electrodynamic vibrators. Trudy LPI Designs and no.192:141-153 '58. (Vibrators) (MIRA 11:6)

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VASYUFINSKIY, S. B., MAGAYENKO, G. P.

"Design Diagrams and Basic Equations of Electrodynamic Vibrators,"

(Dynamics and Strength of Machines; Collection of Articles) Moscov, Mashgiz, 1958. (Series: Its: Leningrad Polytech Inst. Trudy, No. 192) 234 p.

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SOY/105-59-3-8/27 8(2) Vasyutinckiy, S. E., Candidate of Pechnical Sciences, AUTHORS: Nagayenko, G. P., Engineer, Fedorishin, M. I., Engineer Shielding Solid Steel Parts of a Vibration Test Stand Against TITLE: Alternating Magnetic Fields (Ekranirovaniye stal'nykh massivov vibrostenda ot peremennogo magnitnogo polya) Elektrichestvo, 1959, Nr 3, pp 37 - 41 (USSR) PERIODICAL: This is a description of an electrodynamic vibration test ABSTRACT: stand. If in such a test stand copper shields in the form of short-circuited rings are installed, which cover the surface of the core of the annular pole in those places, where the movable winding is located, this will lead to a considerable reduction of the active and reactive power output. No reduction, however, of the mechanical force developed by the test stand will occur due to this measure. By a table it is shown that the application of chields leads to a reduction of the power drawn by the test stand from the supply grid by a factor of 4.23 and to a reduction of the active power in the iron by a factor of 11.5. The copper losses in the movable winding have been reduced by the application of shields. Card 1/3

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Shielding Solid Steel Parts of a Vibration Test Stand 500/105-59-3-8/27 Against Alternating Magnetic Fields

The calculation of apparatus similar to test stands without shields is very complicated. It can be simplified if it is assumed that if shields are used the magnetic alternating field does not enter the iron core. This assumption has been proved right by experiment. It was shown that with chields the magnetic alternating flux at the core surface is generated by a magnetizing force, which is only 5-10% of the magnetizing force of the movable winding. In this case a system composed of a movable alternating current winding and of chert-circuited shields can with sufficient accuracy be looked upon as an air transformer with three windings and chort-circuited secondary windings. Equations (1) (2) and (2) are written down, specifying the EMF of the windings. The solutions of these equations give the voltage at the movable winding and the amperages in the shields. The active power is computed according to equation (7), and the intrinsic losses in the movable winding according to equation (C). In order to check this method of computing the shields and of estimating the efficiency of shielding experiments were carried out at a model test stand of the Leningradskiy politekhnicheskiy

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Shielding Solid Steel Parts of a Vibration Test Stand 507/105-59-3-8/27 Against Alternating Magnetic Fields

> institut (Leningred Polytechnical Institute). The results obtained and the results advanced in this paper demonstrated that a shielding of iron cores by copper shields in installations as described here is very effective. A shielding of iron cores is particularly expedient for high frequency. The experiments carried out substantiated the correctness of the method advanced in this paper of the electromagnetic calculation of apparatus, which are similar to the electrodynamical vibration test stand and which are equipped with a magnetic shielding against magnetic alternating fields. There are 7 figures, 1 table and 3 Soviet references.

ASSOCIATION: Leningradskiy politekhnicheskiy institut im. Kalinina (Leningrad Polytechnical Institute imeni Kalinin)

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VASYUTINSKIY S.B.

PHASE I BOOK EXPLOITATION SOV/4917

- Piotrovskiy, Lyudvig Marianovich [Deceased], Svyatoslav Borisovich Vasyutinaky, and Elena Dmitriyevna Nesgovorova
- Ispytaniye elektricheskikh mashin. Chast' 2: Transformatory i asinkhronnyye mashiny (Testing Electric Machinery. Pt. 2: Transformers and Induction Machines) Moscow, Gosenergoizdat, 1960. 290 p. Errata slip inserted.

Ed.: A.S. Usser; Tech. Ed.: O.S. Zhitnikova.

- PURPOSE: This book is a teaching aid for students working in electrical laboratories in power-engineering and electrical-engineering schools of higher education. It can also be of use to electrical engineers concerned with the testing of electrical machines.
- COVERAGE: The book covers the testing of transformers and "collectorless" induction machines of various types for diverse operating conditions. Related general problems are presented. The manual also includes a description of the industrial testing of these machines carried out in accordance with the All-Union State Standards (GOST) now in force. The first section of the book was

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sov/4917 Testing Electric Machinery (Cont.) written by L.M. Piotrovskiy and S.B. Vasyutinskiy; the second by L.M. Piotrovskiy and Ye.D. Nesgovorova. The authors thank A.I. Vazhnov, P.Yu. Kaazik, and M.I. Fedorishin. There are 44 references: 38 Soviet, 3 German, and 3 English. TABLE OF CONTENTS: SECTION I. TESTING OF TRANSFORMERS Ch. I. General Problems Related to the Testing of Transformers 11 11 1. Educational testing of transformers 12 Industrial testing of transformers
Name plate of a transformer 13 13 4. Basic definitions 14 Symbols
Methods of testing transformers 15 16 7. Methods of experimenting and processing of the results 18 8. Basic equations of two-winding transformers 19 Ch. II. D-C Measurement of Winding Resistances in Transformers 19 1. Methods and conditions of d-c measurements f resistances 22 2. Reduction of resistances to rated temperature Card-2/12 <u>. 19</u>14

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VASYUTINSKIY, S.B., inzh.; NAGAYENKO, G.P., inzh.

Determination of the principal dimensions of electrodynamic vibrators. Vest. elektroprom, 32 no.10:73-75 0 '61. (MIPA 14:9) (Vibrators)

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VASYUTINSKIY, S.B., kand.tokhn.nouk, dotsont; NAGAYENKO, G.F., jush.

Determination of the impedance of inductively coupled singlelayer windings operating in a wide frequency range. Elektrichestvo no.12:49-52 D '65. (MIRA 18:32)

1. Loningradskiy politekhnicheskiy institut imeni Kalinipa.

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