國際的目標 化合金管理 化合合合合 on street hit in the Britship sector and the sector of the ni er ZAMHAROVA, M.G. Symposium on chemotherapy in pulmonary tuberculosis. Probletub. (MIRA 11:10) 36 no.6:119-122 '58 (TUBERCULOSIS)



CIA-RDP86-00513R001963610011-3

ZAKHAROVA, M.I.; HELYATSKAYA, N.S.

Substructure of crystals of a supersaturated solid solution of silver in aluminum during the decomposition process. Fiz.met.i metalloved. 14 no.5:678-682 N '62. (MIRA 15:12)

1. Moskovskiy gosudarstvennyy universitet im. M.V.Lomonosova. (Solutions, Supersaturated) (Aluminum-silver alloys--Metallography)

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	2/500201	<pre>USSR/Physics - Compression Sep 49 Metals, Solubility (Contd) which is the equilibrium concentration for higher pressure. Submitted by Acad N. T. Gudtsov 4 Jul 49.</pre>	<pre>WESR/Physics - Compression Sep 49 Metals, Solubility "Wariation in the Houndary Solubility of Metals in the Solid State Under the Influence of Pressure During Compression on All Sides," M. I. Sakharova, Phys Faculty, Woscow State U imeni A. V. Lomonosov, 3 pp "Dok Ak Mauk SSSR Vol LXVIII, No 1 Concentration of a solid solution which is the equilibrium concentration for atmospheric pressure varies under action of high pressure during compression on all sides. It takes from 2 to several tens of hours to establish concentration 2/5000101</pre>	

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ZAKHAROVA, M. I. Prof.

"The Influence of Pressure on Phase Transformations in Alloys," a paper given at the All-University Scientific Conference "Lomonosov Lectures", Vest. Mosk. Un., No.8, 1953.

Translation U-7895, 1 Mar 56

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ZAKHAROVA, M. I.		
USSR/Physics		
Card :	1/1	
Authors :	Balli, D., and Zakharova, M. I.	
Title :	Investigation of the structure and properties of Cu-Ni-Fe alloys	
Periodical :	Dok1. AN SSSR, 96, Ed. 4, 737 - 740, June 1954	
Abstract : The structure and properties of ten Cu-Ni-Fe alloys, with varying copper, nickel and iron contents, were investigated. A comparison of the x-ray-analysis results, with the data of the measured coercive force and Curie point, showed that nuclei of new phases and non-equilibrium composition are formed in the basic crystalline lattice, during the initial stages of decomposition of the solid solution. The main role in this period is played by stresses of the third order and the coercive force reaches values of tens of Oerstedts for a majority of the alloys. Two references. Tables, graphs.		
	The M. V. Lomonoscy State University, Moscow, USSR	
	Academician G. V. Kurdyumov, March 4, 1954 Letton B-82533, 2 Feb 55	

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ZAKHAROVA, M.I.

P∆ - 1222

SUBJECTUSSR / PHYSICSCARD 1 / 2AUTHORZACHAROVA, M.I., SOEOLEVA, I.N.TITLEThe Decay of a Solid Solution in Thin Plates.PERIODICALDokl. Akad. Nauk, 108, 841-842 (1956)Publ. 6 / 1956reviewed 8 / 1956

According to N.N.BUJNOV and R.M.LERIMAN, Izv.Akad.Nauk. SSR, Ser.fis.No 3, 363 (1951) the decay of a solid solution probably takes a different course from a certain critical depth onwards than in thick samples. In order to find this out, the temperature dependence of the decay of a solid solution of Cu in Al was investigated in samples of 0,4; 0,02 and 0,004 mm thickness. After quenching, the plate-shaped samples were annealed for one hour at 180,200,220, ... 300°. The decay process was radiographically analyzed from the modification of the lattice constant of the primary solid solution. Not only in samples of 0,004 mm thickness, but also in such as are 0,02 mm thick, decay develops in a different manner than in 0,4 mm samples. Domains with different degrees of oversaturation exist in samples of 0,02 and 0,004 mm thickness after tempering at 260 and 280°. After tempering at 260° (or 280°) crystal domains with a concentration similar to the initial concentration (i.e. with equilibrium concentration) predominate. In the samples of 0,4 mm thickness tempering at 260 and 280° causes homogeneous decay, on which occasion the lattice constant changes steadily. If temperature is further increased, the solid solution has a uniform concentration at all thicknesses from 0,004 to 0,4 mm within the entire domain, and the lattice constant diminishes by the increase of solubility.

APPROVED FOR RELEASE: 09/19/2001

ZAKHAROVA, M.I.; STETSENKO, P.N.

111

Magnetic properties and structure of an Pe - V(27%) alloy. Vest. Mosk. un. Ser. mat., mekh., astron., fiz. khim., 12 no.5: 47-52 '57. (MIRA 11:9)

1.Kafedra magnetisma Moskovskogo gosudarstvennogo universiteta. (Iron-vanadium alloys---Magnetic properties)

APPROVED FOR RELEASE: 09/19/2001

"APPROVED FOR RELEASE: 09/19/2001 CIA-RDP86-00513R001963610011-3
XAXHAROVA, M.I.; STETSERKO, F.N.
Phase transformations in Fe-V alloys. Vest. Mosk. un. Sor. mat.; Phase transformations in Ye-V alloys. Vest. Mosk. un. Sor. mat.; (MIRA 11:9)
J.Kafedra magnetisma Moskovskogo gosudaratvennogo universiteta. (Iron-vanadium alloys--Metallography)

APPROVED FOR RELEASE: 09/19/2001


)RS: Zakharova, M.I. and Khatanova, M.A.

AUTHORS: Zakharova, M.I. and induction of Crystals of the α and σ TITLE: The Mutual Orientation of Crystals of the α and σ Phases on the Decay of the Solid Solution in Alloys of Iron and Vanadium (Vzaimnaya origentirovka kristallov α-Iron and Vanadium (Vzaimnaya origentirovka kristallov αi σ-faz pri raspade tverdogo rastvora v splavakh zheleza s vanadiyem)

PERIODICAL: Kristallografiya, 1958, Vol 3, Nr 3, pp 376 - 378 (USSR)

ABSTRACT: Fe-V alloys at temperatures above 1 234 °C form a solid solution α with a cubic face-centred lattice. Below this temperature the solid solution decays and a σ phase with the β - U structure having 30 atoms per unit cell separates. An alloy of 26% V in Fe was annealed at 975 °C and decayed to the two phases. Monocrystalline specimens, prepared by heating for 60 hours at 1 350 °C and quenching in water were used for X-ray examination. Specimens of 1 cm dia. were thus converted to single crystals and were cut up for examination. Laue phyto-graphs were taken after different annealing times at 975 °C. For times of 1-30 hours no changes were evident. After 40 hours, spots showed that the nuclei of the σ phase were oriented parallel to the 001 plane of the α phase. After Card 1/2 155 hours annealing the orientation was seen to be such that

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HCELE:

 AUTHORS: Zakharova, M.T. and Khatanova, N.A. 70-3-3-29/36 TITLE: The Substructure of Crystals of the γ Solid Solution of Nickel in Iron During Polymorphic Transformation (Sub-Nickel i
an alloy of iron with 520 michae the initial stage proceeds microscopic analysis showed that the initial stage proceeds following the martensitic type of transformation scheme. In following the martensitic type of transformation scheme. In this the layers of the α -phase are oriented parallel to the lli this the layers of the α -phase are oriented parallel to the elastic plane of the γ -phase. Because of the low value of the elastic limit of the Fe-Ni alloys at 400°C, the coherence of the limit of the Fe-Ni alloys at 400°C, the coherence of the stages of the γ - and α -phases is destroyed in the initial lattices of the transformation and the further growth of the stages of the transformation and the further growth of the nuclei of the α -phase proceeds by diffusion. In the matrix nuclei of the nuclei a zone of plastic deformation is formed clearly distinguishable under microscopic investigation. There are 4 figures and 7 references, 1 of which is Soviet, 1 German and 5 English.
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"APPROVED FOR RELEASE: 09/19/2001 CIA-RDP86-00513R001963610011-3 70-3-3-29/36 The Substructure of Crystals of the Y Solid Solution of Nickel in Iron During Polymorphic Transformation Mokovskiy gosudarstvennyy universitet imeni M.V. Lomonosova (Moscow State University imeni M.V. Lomonosov) ASSOCIATION: March 22, 1957 SUBMITTED: Card 2/2

APPROVED FOR RELEASE: 09/19/2001

CIA-RDP86-00513R001963610011-3

SOV/126-6-3-12/32

ESSING IN MARKED AND DESCRIPTION OF THE OWNER

AUTHORS: Zakharova, M. I., Ignatova, I. A. and Khatanova, H. A.

Investigation of the Phase Transformation $\gamma \rightarrow (\gamma + \alpha)$ in Alloys of Iron with Nickel (Issledovallye fazovogo prevrashcheniya $\gamma \rightarrow (\gamma + \alpha)$ v splavakh zheleza s nikelem) TITLE:

PERIODICAL: Fizika Metallov i Metallovedeniye, 1958, Vol 6, Nr 3, pp 475-479 (USSR)

The polymorphous transformations in alloys of iron with 30 and 32% Ni are investigated since in spite of the ABSTRACT:

fact that much work has been done on the problem of $\gamma \rightarrow (\gamma + \alpha)$ transformations in Fe-Ni alloys (Ref 1), the extreme stubility of the non-equilibrium state in these alloys has so far not been satisfactorily clarified. The alloys were produced from electrolytic iron and electrolytic nickel. After casting, the alloys were subjected to homogenization annealing at 1000°C for ten hours, then to heating for 18 hours at 600°C which was followed by quenching in water. The single crystals were produced by the method of recrystallisation at 1200°C; after continuous annealing for 60 hours, crystals of 20 mm² grew in 1 mm thick plates. The investigations were effected by X-ray Card 1/3 and microscopic analysis of polycrystalline specimens and

all in this was shown and so the state of the second state of the

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CIA-RDP86-00513R001963610011-3

SOV/126-6-3-12/32 Investigation of the Phase Transformations $\gamma \rightarrow (\gamma + \alpha)$ in Alloys of Iron with Nickel

X-ray analysis of static single crystals. The process of polymorphous transformation was studied for isothermal heating at a temperature of 400°C; at this temperature a 32% Ni containing alloy in the equilibrium state should contain about 27% of the α -phase. For investigating the $\gamma \rightarrow (\gamma + \alpha)$ transformation by X-ray structural analysis, powder was filed from the homogenized specimen which was heated at 600°C for 20 hours and then subjected to isothermal annealing at 400°C. The X-ray patterns were photographed using iron radiation in cameras of 114 cm dia; the specimen dia, equalled 0.4 mm. It was established that at 400°C the transformation is very slow. Deformation of the alloys at room temperature does not only accelerate the process of γ to α transformation; deformation of an alloy after being subjected to martensite transformation at -196°C will accelerate also the reverse γ to α transformation. At temperatures above the martensitic point, the initial stage of the γ to α transformation proceeds Card 2/3 according to the relations governing the reconstruction of

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SOV/126-6-3-12/32 Investigation of the Phase Transformation $\gamma \rightarrow (\gamma + \alpha)$ in Alloys of Iron with Nickel the lattice in the case of martonsitic transformations. The forming inter-layer of the γ -phase is located parallel to the plane (111) of the γ -phase. Apparently for a tempering temperature of 400°C the lattice coherence is disturbed in the initial stage of transformation, which brings about a braking of the transformation process. Further increase in the growth of the nuclei of the α -phase is by diffusion; deformation zones are formed in the matrix around the nuclei. There are 2 figures, 1 table and 4 references, 2 of which are Soviet, 2 English. ASSOCIATION: Moskovskiy gosudarstvennyy universitet imeni M. V. Lomonosova (Moscow State University imeni M. V. Lomonosov) SUBMITTED: June 23, 1956 (initially), Feb.8, 1957 (after revision). 1. Iron-nickel alloys--Transformations 2. Iron-nickel alloys --Stability 3. Iron-nickel alloys--Casting 4. Iron-nickel alloys --Heat treatment 5. Iron-nickel alloys--X-ray analysis Card 3/3

APPROVED FOR RELEASE: 09/19/2001

ZAKHAROVA, M.I.; KHATANOVA, N.A.

Investigation of structural changes in Ne-Ni alloys during the polymorphic $\chi \rightarrow \chi + c$ transformation. Isv. AN SSSR. Ser. fiz. 22 no.10:173-176 U '58. (MIRA 12:3)

1. Moskovskiy gosudarstvennyy universitete im. M.V. Lomonosova. (Iron-nickel alloys)

APPROVED FOR RELEASE: 09/19/2001



TITLE: An Investigation of the Phase Composition of Tron-tanatrom and Iron-Chromium Alloys (Issledovaniye fazovogo sostava splavov zheleza, s vanadiyem i zheleza s khromom) PERIODICAL: Doklady Akademii Nauk SSSR, 1958, Vol. 119, Nr 3, pp. 498-500 (USSR) ABSTRACT: Though there is a domain of the σ -phase in the state diagrams of the alloys in question which passes over into diagrams of the alloys in question which passes over into for Fe-V-alloys and at 820°C for Fe-Cr-alloys, these trans- for Fe-V-alloys are transformed rapidly in the single-phase these alloys are transformed rapidly in the single-phase region of the σ - as well as of the α -phase. Thus the brittleness occurs very obviously after annealing at 400-550°C in these alloys that belong to the single-phase region. The plasticity is here reduced to zero, by this region. The plasticity is here reduced to zero, by this		20-119-3-27/65
and Iron-Chromium Alloyd (m i sheleza s khromom) splavov zheleza, s vanadiyem i sheleza s khromom) PERIODICAL: Doklady Akademii Nauk SSSR, 1958, Vol. 119, Nr 3, pp. 498-500 (USSR) ABSTRACT: Though there is a domain of the σ -phase in the state diagrams of the alloys in question which passes over into diagrams of the alloys in question which passes over into the domain of sclid solutions of the α -phase at >1234°C the domain of sclid solutions of the α -phase at >1234°C the domain of sclid solutions of the α -phase at >1234°C the domain are assumed to be more complicated, because formations are assumed to be more complicated, because these alloys are transformed rapidly in the single-phase these alloys are transformed rapidly of the single-phase 400-550°C in these alloys that belong to the single-phase 400-550°C in these alloys that belong to the single-phase region. The plasticity is here reduced to zero, by this region. The plasticity is here reduced to zero, by this	AUTHORS:	Zakharova, M. I., Ignatova, I. A., 20-119-9-21/09 Semenova, L. A., Khatanova, N. A.
PERIODICAL: Doklady Akademii Nauk SSSR, 1958, Vol. 119, Nr 9, pp. 498-500 (USSR) ABSTRACT: Though there is a domain of the σ -phase in the state diagrams of the alloys in question which passes over into the domain of solid solutions of the α -phase at >1234°C the domain of solid solutions of the α -phase at >1234°C for Fe-V-alloys and at 820°C for Fe-Cr-alloys, these trans- formations are assumed to be more complicated, because these alloys are transformed rapidly in the single-phase these alloys are transformed rapidly in the single-phase these of the σ - as well as of the α -phase. Thus the brittleness occurs very obviously after annealing at 400-550°C in these alloys that belong to the single-phase region. The plasticity is here reduced to zero, by this	TITLE:	and Iron-Chromium Alloyd (sheleza s khromom)
ABSTRACT: Though there is a domain of the σ -phase in the state diagrams of the alloys in question which passes over into the domain of solid solutions of the α -phase at >1234°C the domain of solid solutions of the α -phase at >1234°C for Fe-V-alloys and at 820°C for Fe-Cr-alloys, these trans- formations are assumed to be more complicated, because these alloys are transformed rapidly in the single-phase these alloys are transformed rapidly in the single-phase these of the σ - as well as of the α -phase. Thus the brittleness occurs very obviously after annealing at 300-550°C in these alloys that belong to the single-phase region. The plasticity is here reduced to zero, by this index practical applicability is restricted. According to	PERIODICAL:	Doklady Akademii Nauk SSSR, 1958, Vol. 119, Nr 2, 498-500 (USSR)
		Though there is a domain of the σ -phase in the state diagrams of the alloys in question which passes over into the domain of solid solutions of the α -phase at $> 1234^{\circ}C$ for Fe-V-alloys and at 820°C for Fe-Cr-alloys, these trans- formations are assumed to be more complicated, because these alloys are transformed rapidly in the single-phase region of the σ - as well as of the α -phase. Thus the brittleness occurs very obviously after annealing at 0.5500C in these alloys that belong to the single-phase
Card 1/4 assumed to precipient	Card 1/4	agsumed to preserve

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An Investigation of the Phase Composition of Iron-Vanadium 20-119-3-27/65 and Iron-Chromium Alloys

> modification of the properties which does not correspond to the single-phase structure of the solid solution was observed also above the transformation temperature from σ - into the α -phase (ref. 6,7). It is difficult to be explained by the atomic regulation which is assumed at low as well as at high temperatures by several authors (ref 7). In the present paper the structure of the alloys in question was to be investigated after a heating between 1400 and 600°C with quenching in water. The investigation was carried out by means of X-ray diffraction methods in the polycrystal and by means of microscopical analysis. The alloys were homogenized after casting at 1300°C from 20 to 100 hours and immediately afterwards quenched in water. Structure of the iron-vanadium-alloys. The radiographs of the powder obtained by means of a file were taken with a chromium radiation. After a homogenization at 1300°C these alloys are (with a vanadium content of 28,5-74 %) not single-phase, but two-phase. It was proved microscopically that on a background of the crystals of the α -phase

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An Investigation of the Phase Composition of Iron-Vanadium and Iron-Chromium Alloys 20-119-3-27/65

(hardness ~ 250 kg/mm²) crystals of another phase with a hardness three times greater than the first mentioned become visible. The content of this last phase increases with increasing vanadium content. The radiograph confirmed this: 2 systems of lines appear on it. The other phase is denoted as β -phase by the authors. The content of the phases was determined in the case of different vanadium contents. The two phases still existed at temperatures above 1150°C. In the case of annealing at 800°C the alloy with V-content of 28,5 % consists of the α -phase only. From 43 % V on it consists of α - and Y-phase. In the case of annealing at 600:C and 49,5 % V it consists of the σ -phase only. Thus the course of the phase transformations in more complicated at a vanadium content of 28,5-74 % between 1400 and 600°C, than described by the phase diagram in publications, i.e.

 $(\alpha + \beta) \longrightarrow \alpha \longrightarrow (\alpha + \sigma) \longrightarrow (\alpha + \sigma)$

Chromium-iron-alloys. After the same treatment the

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:	An Investiga Vanadium and	ation of the Phase Composition of Iron- 20-119-3-27/65 I Iron-Chromium Alloys
		microscopical and radiographic investigation showed that the alloys with 35, 42 and 48 $\%$ Cr consist of the α - and β -phase crystals in the case of annealing at 1300°C. The amount of the β -phase decreases with dropping temperature (figure 1,2). In the chromium-iron-alloys with 35-48 $\%$ Cr the phase transformations consist of a polymorphous transformation of the O - into the α -phase as well as of the α - into the β -phase, exactly as it was the case with the above mentioned vanadium. There are 3 figures and 7 references, 2 of which are Soviet.
	ASSOCIATION:	Moskovskiy gosudarstvennyy universitet im. M. V. Lomonosova (Moscow State University imeni M. V. Lomonosov)
	PRESENTED:	November 12, 1957, by A. A. Bochvar, Member, Academy of Sciences, USSR
	SUBMITTED:	November 12, 1957
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Phase Transformations in the System Iron-Vanadium	
conditions for the formation of this new (β) phase. The quantity of a-phase in the 47.7 and 27% V alloy was found by Nechvolodov's method to be 35 and 10% respectively. The magnetic properties of the low- and	
high-vanadium alloys annealed at 1350 °C for 60 hours are shown as functions of temperature in Figs 3 and 4,	•
respectively. The work shows that there are two polymorphic changes (β -a and a-g) in the Fe-V alloys, both proceeding slowly in the 1000-1300 °C range. The β -phase has a Curie point of about 200 °C and	
Card crystallizes in a cubic face-centered lattice. 2/2 There are 4 figures.	
ASSOCIATION: Otdeleniye stroyeniya veshchestva fizicheskogo fakul'teta MGU (Structure of Matter Department, Faculty of Physics, MGU)	
SUBMITTED: March 28, 1958	4
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	7. A	K	++-*	९७	v۸,	M-1	-T.,		1	263	266	E	280	752			•				
		ACT SOUND SOUND SOUND SOUND SOUV 3355	autry Akademiya nauk SSSR. Institut matallurgii. Mauchnyy Aovet Po problema sharoprochnych splarov	<pre>Imaledowaniye po zharoprochnym splavam, t. IV (Studies on Heat-fa- maledowaniye yol. 4), Mascew, Izd-vo AN SSSR, 1959. 400 p. Erska alip inderted. 2,200 copies printed.</pre>	24. of Publishing House: Y. A. Klisov; Tech. Ed.: A. P. Ouseva; Editorial Board: I. P. Bardin, Academician; Q. Y. Kurdyusov, Academician; W. V. Agery: Corresponding Member, USSR Academy of Academician; M. V. Agery: Corresponding Member, USSR Academy of Sciencesi I. A. Oding, I. M. Pavlov, and I. P. Cudin, Candidate of Perhonal Sciences.	Fundods: This book is intended for matallurgists concerned with the structured settliurgy of alloys.	COVERAGE: This is a collection of specialized studies of various problems in the structural metallurgy of heat-medistant alloys you are concerned with transmitted principles, some with des- down are concerned with transmitted principles, some with properties	Criptions on the way suprementation of the second occurring under of specific anterials. Varices principal reported on . For details, specified controls are studied and reported on . For details, see Table of Content. The studied and recordsmidd by a num- her of references. Doils GordsLand root-Goviet.		Zatharora. L. T. N. H. JERIGTA, L. N. Samonora, and - K. ENTURNY: Intestigation of Phate Transformations 4. Computations and Iron-Chronium Alloys	.29	X Mumprith O. A., and I. P. Zudin. Eigh Temperature Creep Stemmeth of Complex Alloys of Verrite with Chronium, Wandidue, Thatesten, and Malydenum	" Pridentaev. Mrrw. Some Problems in the Theory of Meat	Adding, I. A., and V. <u>F. Caminov. Man</u> Wethod of Extrapolating <u>- English</u> Strength Properties from Short-time Endurance Test with	Stanyukovich, A. Y. Investigation of Plasticity Properties Card 9/12	1. · · · · · · · · · · · · · · · · · · ·			۹.		
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SOV/48-23-5-23/31 18'(7) Zakharova, M. I., Mogarycheva, I. B. AUTHORS: Investigation of the Phase Transformations in Copper - Tin Alloys (Issledovaniye fasovykh prevrashcheniy v splavakh TITLE: med'-olovo) Izvestiya Akademii nauk SSSR. Seriya fizicheskaya, 1959, Vol 23, PERIODICAL: . Nr 5, pp 643 - 645 (USSR) It is mentioned by way of an introduction that several earlier investigations had dealt with the decomposition of oversaturat-ABSTRACT: ed solid solutions. The subject of the present paper is the eu-. . tectic transformation and the phase transformation $\beta \rightarrow \beta + \alpha$. Reference is then made to two papers by Isaichev and Kurdyumov concerning the disordered position of atoms at room temperature in the β phase, and the ordered position of atoms at 700°C, with 25 - 28% tin. The investigation under review deals with copper alloys with 25.5%, 27.5% and 30.5% tin. The samples are monocrystals which are investigated immediately after annealing at 700°C. In addition, a general investigation was made of the copper alloys with 32.6% tin, and the alloy with 27.8% tin was investigated with regard to the sutectic transformation at Card 1/3

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Investigation of the Phase Transformations in Copper - SOV/48-23-5-23/31 Tin Alloys

350 and 400°C. An alloy with 25.5% tin was used to investigate

the phase transformation $\beta \rightarrow \beta + \alpha$ at 550°C. Investigation methods applied were the diffraction of X-rays in monocrystals, the crystal vibration and the monochromatic emission of molybdenum. The results are explained on the strength of roentgenograms and Laue diagrams. Ficture (Fig 1) shows the beginning of separation of the β phase in the alloy with 25.5% tin. Also the de-composition of the β phase in the alloy with 30.5% tin is dealt with. In these investigations, the phase transformations are inferred from the location of the diffraction maxima. For example, the diffraction pictures (Figs 2 and 4) of the alloy with 27.8% tin, taken at various time intervals after the thermal treatment, are shown, and the progressive phase transformation is investigated thereon. The sutectic transformation is investigated in the same way and described with a number of pictures. There are 7 figures and 13 references, 6 of which are Soviet.

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ZAKHAROVA, M. I., VAN CHYA-FOY

High Manganese Steel and Aluminium Zinc Alloys."

Moscow State Univ., Faculty of Physics, Moscow, USSR.

paper submitted for 5th Gen. Assembly, Symposium on Lattice Defects, Intl. Union of Crystallography, Cambridge U.K. Aug 1960.

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Conditions for the formation of the sigme phase in alloys. Trudy Sek.metalloved.i term.obr.met.MTO mash.prom. no.2:39-51 '60. (MIRA 14:4) (alloys-Metallography) (Phase rule and equilibrium)

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ZAKHAROVA, M.I., doktor fiz.-mat.nauk, prof.

Structural transformations in highly coercive alloys. Trudy Sek. metalloved.i term.ob. met.NTO mash.prom. no.2:52-58 :60. (MIRA 14:4)

(Alloys-Magnetic properties) (Phase rule and equilibrium)

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S/180/60/000/005/020/033 Ell1/E135 Threstigation of Substructure in the Decomposition of the γ-Solid manganese steel the austenite in carbon-impoverished regions is plastic deformation of unchanged regions of austenite and conversion of a single into a polycrystal. In the high-manganese stage of ageing. There are 2 figures and 4 references: 3 Soviet and 1 in Acta Crystallografica. SUEMITTED: February 9, 1960 Card 2/2

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s/126/60/009/02/013/033

Investigation of the Decomposition of Super-Saturated Solid Solution in Manganese Steel

manganesc-atom movement in the austenite lattice. For studying in detail structural changes in the first stage of decomposition X-ray diffraction with mixed and monochromatic radiation was used on single crystals (prepared by recrystallization of 5% elongated specimens) at 1 100 C in evacuated quartz tubes. A special holder fixed the crystal to the goniometric head. Figure 1 shows the diffraction pattern from a hardened single crystal; Figures 2-4 those from single crystals tempered at 750 °C for 2, 20 and 180 min, respectively. Figure 5 shows individual regions of a series of patterns obtained when the crystal tempered at 750 °C for 2 min obtained when the crystal tempered at 750 $^\circ$ C for 2 min was rotated through 1.5 - 3.0 up to 25 . Some of the patterns include small bow-shaped lines and the authors discuss these in terms of the reciprocal lattice (Figures 6, 7). On the basis of their analysis of the

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68626 S/126/60/009/02/013/033 Investigation of the Decomposition of Super-Saturated Solid Solution in Manganese Steel geometry of diffraction patterns they consider nucleation and block effects: disorientation of blocks of the initial solid solution does not increase continuously with increasing size of crystals of the precipitating phase but decreases after reaching a maximum. There are 7 figures. ASSOCIATION: Moskovskiy gosudarstvennyy universitet imeni M.V. Lomonosova (Moscow State University imeni M.V. Lomonosov) SUBMITTED: July 22, 1959 Card 3/3 tirfitter set and a start were

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承住现住行行业推销经济动作了资源共产的通过知道的标志设计与在规范的环境的运送资料中多些 ZAKHAROVA, MI. s/126/60/010/01/008/019 E111/E335 Zakharova, M.I. and Van Khua-Fou Investigation of Eutectoidal Transformation¹⁸in AUTHORS: TITLE: 1460 Austenitic Steels, 8 Fizika metallov i metallovedeniye, 1950, Vol.10, PERIODICAL: No. 1, pp 70 - 74 The authors maintain that in spite of the numerous investigations of the eutectoidal transformation its initial TEXT: stages need further study. They report their work on the transformation in a steel with 12% Mn and 1.2% C and one with 1.77% Mn and 2% C, using X-ray analysis of poly- and single crystals. For the 12% Mn steel the investigation was carried out after tempering at 670°C. The authors discuss the patterns obtained and calculate stresses, crystal size and blattice deformations. Fig. 1 shows the pattern from a single crystal after hardening and tempering for 3 minutes; Figs. 2 and 3 after tempering for 14 and 40 minutes, respectively. A conclusion from the results is that the sequence of alpha. phase liberation is different at the same temperature in a Card 1/2

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investigated at 500 C on the initial s impoverished	n of Eutectoidal Tr and a single crysta before it in the la after tempering for polycrystalline sp tages redistributi areas the martensi There are 3 figur Moskovskiy gosud M.V. Lomonosova M.V. Lomonosov) February 11, 1960	ransformation j al; after carb atter. The 1.7 or different ti becimens. The on of carbon o te point is ab es and 2 Sovie arstvennyy univ (Moscow State	bide liberation 7% Mn steel was mes at 700 °C a work shows that ccurs; for car ove room t references.	teels in s and t in rbon-	
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s/126/60/010/004/009/023 E021/E406 **AUTHORS:** VI Zakharova, M.I. and Yeliseyeva, I.I. νl Study of the Initial Stages of Ageing of Aluminium-Zinc TITLE: Alloys `PERIODICAL: Fizika metallov i metallovedeniye, 1960, Vol.10, No.4, pp.560-563 The ageing of aluminium-zinc alloys of single and polycrystals TEXT : was investigated by X-ray analysis and by etch figure techniques. The starting materials were AB 000 (AV000) aluminium and 99.8% pure zinc. The single crystals were prepared by slowly cooling melts of Fig.1 shows an X-ray photograph of a single crystal of the alloy. aluminium - 10% zinc alloy after natural ageing. The presence of streaks indicates the formation of areas rich in zinc. Single crystals and coarse grained alloys were also polished electrolytically and chemically etched. Etch figures increased with increase of zinc content from 5 to 15%. After two days ageing at room temperature, the etch-figures were uniformly distributed within the grains. After five days natural ageing in individual crystals of the aluminium-10% zinc alloy, the etch figures were arranged in parallel lines (Fig.2). The uniformity of the etch Card 1/2

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CIA-RDP86-00513R001963610011-3

s/126/60/010/004/009/023 E021/E406 Study of the Initial Stages of Ageing of Aluminium-Zinc Alloys figures depended on the orientation of the grains and the time of ageing. After seven months ageing, the/etch figures were distributed uniformly in all the grains forming a network with an angle of 70° (Fig.3). After 50 hours ageing at 150°C, spots appear on the Debye rings corresponding to (311) reflections. 255 hours, the intensity of these spots sharply increases and After streaks appear in a radial direction across the Laue maxima. After 320 hours the intensity and angular length sharply decreases This effect is caused by the reorientation of small volumes of the matrix as a result of differences in the specific volumes of the matrix and the precipitating planes. 5 figures and 2 Soviet references. There are ASSOCIATION: Moskovskiy gosudarstvennyy universitet imeni M.V.Lomonosova (Moscow State University imeni M.V.Lomonosov) SUBMITTED: February 11, 1960 Card 2/2

APPROVED FOR RELEASE: 09/19/2001

195

	SOURCE CODE: UR/0363/67/003/001/0087/009
AUTHOR: Prokoshkin, D.	
ORG: Notallurgy Institu notallurgii Akademii nau	te im. A. A. Baykov, Academy of Sciences, SSSR (Institut k SSSR)
TITLE: Isothermal section has diagram	ons at 600 and 750 °C of the molybdenum-titanium-zirconium
OURCE: AN SSSR. Izvest	lya. Neorganicheskiyo materialy, v. 3, no. 1, 1967, 87-93
OPIC TAGS: molybdenum a	alloy, zirconium alloy, titanium alloy, alloy phase diagram
hermal sections at these em were constructed. A rom the Mo-Ti system to tate of the alloys on th see Fig. 1). The region f the concentration tria mmixing of the β solid so ontent of Mo in Zr and β	of x-ray and microstructural analyses and measurements of the quenching from the equilibrium state at 750 and 600 °C, iso- two temperatures of the phase diagram of the Mo-Ti-Zr sys- sizable region of a β solid solution, extending continuously the Ti-Zr system and bounded by a region of heterogeneous e side of the Mo-Zr system, was found in the section at 600 of heterogeneous state of the alloys occupies a small part ngle and protrudes toward the titanium corner (see Fig. 1). olution into two solid solutions occurs at an equiatomic 1 at. \$ Ti. Two three-phase regions, $\beta_1 + \beta_2 + \delta$ and $\alpha +$ sterogeneous region. The δ phase extends up to 13 at. \$ Ti
rd 1/3	UDC: 546-3-19-77-821-831

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ACC NR: AP7006204 at 33.3 at. 3 Zr in the ternary system; no approciable solubility has been noted in this phase. The isothermal section at 750 °C (see Fig. 2) is basically no different ably both at the expense of the heterogeneous region (on the side of the Mo-Zr system) of the section. Unimixing of the β solid solution takes place at 57 at. β Ti and at 600 °C) and exists in this section up to 15 at. β at 33.3 at. 4 Zr. Orig. art. SUB CODE: 007/ SUEM DATE: 09Feb66/ ORIG REF: 004/ OTH REF: 008

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"APPROVED FOR RELEASE: 09/19/2001 CIA-RDP86-00513R001963610011-3
"AKHAROVA, M.I. (Moskva); PROKOSHKIN, D.A. (Moskva)
Investigating the system niobium - molydenum - chromium.
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Izv. AN SSSR. Otd. tekh. nauk. Met. i topl. no.4:59-67
(MIRA 14:8)
J1-Ag '61.
(Niobium-molybdenum-chromium alloys--Metallography)
(Phase rule and equilibrium)

APPROVED FOR RELEASE: 09/19/2001

CIA-RDP86-00513R001963610011-3

s/180/62/000/006/015/022 E071/E151 Zakharova, M.I., and Mogarycheva, I.B. (MOBCOW) **AUTHORS**: Ageing of a copper-tin eutectic alloy TITLE: Izvestiya. Otdeleniye PERIODICAL: Akademiya nauk SSSR. tekhnicheskikh nauk. Metallurgiya i toplivo, no.6, 1962, 147-149. An investigation was made of the structure of single crystals of alloys of copper with 27.8 and 25.5 wt.% of tin, TEXT: together with hardness determinations on polycrystalline specimens (27.8 wt.% Sn) both after hardening and during natural ageing. The microhardness of the polycrystalline specimens increased from 200 to 450 kg/mm² during two years of ageing. To elucidate structural changes causing this increase in hardness, three single crystals with 27.8 wt.% Sn and one with 25.5 wt.% Sn were examined after ageing for 40 days, 8 months and 3 years. The single crystals were prepared by a slow crystallisation from the melt followed by a homogenising treatment at 600 °C for 26 hours. Mixed and monochromatic Mo radiation were used for the X-ray The results obtained indicated that during natural studies. Card 1/2

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Ageing of a copper-	-tin eutectic alloy	S/180/62/000/00 E071/E151	6/015/022
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s/659/62/008/000/002/028 1048/1248

Zakharova, M.I., Mogarycheva, I.B., and Khatanova, N.A. AUTHORS:

Structure of the matrix during the initial stages of de-TITLE: composition of the solid solution

Akademiya nauk SSSR. Institut metalurgii, Issledovania SOURCE: po zharoprochnym splavam. v.8. 1962. 27-31

TEXT: X-ray and microscopic examinations of various Al alloys and Mn steel during the initial stages of decomposition show that at 218°C of the Al-1.25% Si solid solution there is a generated stress not relieved by thermal relaxation, and the matrix is subject to plastic deformation. This is exhibited on the X-ray diagram by asterism and fragmentation of the Laue maxima for the solid solution. The same alloy, annealed for 10 minutes at 2189, shows slip bands under the microscope, and disintegration of monocrystals into smaller structural blocks. Two slip-band systems, intersecting with each other at a 70° angle are observed under certain conditions. Essentially the same microstructure is observed in an Al -

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S/659/62/008/000/002/028 1048/1248

Structure of the matrix during the initial...

10% Zn alloy after natural aging for 7 months, and in steel containing 12% Mn and 1.2%C after annealing for 5 sec. at 670C; electrochemical etching shows that the nature of the microstructure remains unchanged to a considerable depth within the alloy. As all three alloys mentioned have an f.c.c. lattice, the slip plane being (111), it is assumed that the appearance of two slip-band systems intersecting at 70°C is associated with nucleation on the (111) and (111) planes. There are 3 figures.

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CIA-RDP86-00513R001963610011-3

s/659/62/008/000/009/028 1048/1248

SUPERIOR DESIGNATION STATES AND DESCRIPTION OF STATES

AUTHORS: Prokoshkin, D.a., and Zakharova, M.I.

TITLE: The isothermal section at 1200°C of the phase diagram for the system niobium-molybdenum-chromium

SOURCE: Akademiya nauk SSSR. Institut metallurgii, Issledovaniya po zharoprochnym splavam. v.8. 1962. 70-74

TEXT: Alloys of the niobium-molybdenum-chromium system were tempered at 1200° and subjected to a series of microstructure, x-ray, and hardness studies; the results are summarized in the form of the isothermal section at 1200°, and of graphs showing the variations in the lattice parameters of the various phases as a function of the Cr content. The solubility of chromium in niobium at 1200° is 11% (all percentages given are atomic), that of Nb in

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 $\begin{array}{c} S/659/62/008/000/009/028\\ IO48/I248\\ \hline \\ \mbox{The isothermal section...}\\ \hline \\ \mbox{Tr is 2\%. In the binary Mo-Cr system, all alloys containing 10-15\%\\ No consist of a single phase with a b.c.c. lattice. In the ternary$ system Nb-Mo-Cr, all alloys containing above 50% Mo are composed ofanother single phase with a narrow range of homogeneity is confinedwithin the points 62-68% Cr on the 0% Mo line and 11% Mo on the 61%of line; the structure of the phase corresponds to that of the $The J and <math>\beta$ phases exist in the Nb-rich and the Cr-rich corners of and one three-phase regions: $d + \beta, d + \delta, \beta + \delta, and d + \beta + \delta$. There are 4 figures. $d = 2/\delta - 2$

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CIA-RDP86-00513R001963610011-3

S/126/62/014/004/012/017 E193/E383 Zakharova, M.I. and Amosov, Ye.M. A study of the transformation of the β-phase in the copper-beryllium system

PERIODICAL: Fizika metallov i metallovedeniye, v. 14, no. 4, 1962, 559 - 563

TEXT: The object of the present investigation was to study solid-state transformations in the 9.34% beryllium-copper alloy by X-ray and metallographic analysis. Both polycrystalline and single-crystal specimens were used. The results are summarized below. 1) The β -phase, stable at 855 - 890 °C, could not be retained by quenching. Polycrystalline specimens, held at 870 °C for 5 hours and water-quenched, consisted of the Y-phase with a lattice parameter of 2.718 Å. On subsequent ageing at 500 °C the α -phase was formed, the intensity of the X-ray lines produced by this phase increasing as the ageing time increased from 3 min to 7 hours. Examination of microsections revealed that the α -phase particles were formed first at the grain boundaries; after 1-hour ageing at 500 °C the α -phase precipitates could be

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TITLE:

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CIA-RDP86-00513R001963610011-3

ł S/126/62/014/004/012/017 A study of E193/E383 observed in the interior of the grains along the slip lines formed as a result of plastic deformation caused by quenching stresses. 2) Single-crystal specimens, prepared by slow (10 °C/h) cooling of the melt in the crucible and quenched (with the crucible) on reaching 870 °C, had a structure which depended on the rate of cooling during quenching. Specimens quenched in porcelain crucibles consisted of the Y-phase; those quenched in a graphite crucible constituted single crystals of a metastable phase with a face-centred cubic lattice; air-cooling of a single crystal produced by the pulling-out technique resulted in the γ -phase, in which the process of precipitation of the α -phase had 3) Slip on the (110) and (112) planes took place in water-quenched, single-crystal specimens; This was accompanied by the formation of atom aggregates with destroyed periodicity which, on subsequent ageing, became crystals of the α -phase, oriented in accordance with the principle of structural conformity. There are 7 figures. ASSOCIATION: Moskovskiy gosudarstvennyy universitet im. M.V. Lomonosova (Moscow State University im.M.V.Lomonosov) SUBMITTED: Card 2/2 in the second

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CIA-RDP86-00513R001963610011-3

. 35595 s/048/62/026/003/003/015 B139/B104

19.1.210
AUTHORS: Zakharova, M. I., and Khatanova, N. A.
TITLE: Investigation of the structure of solid solutions dependent on crystallization conditions and heat treatment on crystallization conditions and heat treatment
PERIODICAL: Akademiya nauk SSSR. Izvestiya. Seriya fizicheskaya, v. 26, no. 3, 1962, 345 - 348
TEXT: The change in the block structure of the matrices with phase transformations of Al-Si and Al-Cu alloys was investigated. After being hardened at 550°C, single crystals of an alloy of Al with 1.2% Si were tampered at 218°C. The lattice constant changed from 4.0380 to .
A after 10 minutes; and the block boundaries were clearly tampered by an X-ray reflection method. Single crystal plates with an area of 1 - 2 cm⁻¹. The alloy had a dendrite structure immediately after of 10 degree.cm⁻¹. The alloy had a dendrite structure immediately after ielatively few blocks. After 2 min annealing at 280°C the maxima halve, Card 1/3

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B139/B104

Investigation of the structure...

showing that the blocks are disorientated by 4'. The disorientation after 5 min is 6'. After an annealing at 280°C for 10 min the blocks turn, and after 20 min the orientation of the blocks in a crystal increases. Consequently, the formation of the second phase from a solid solution of Si in Al causes a disorientation of the blocks. The structural changes are irreversible. Al alloys with 4 per cent by volume of Cu have a band structure under the same crystallization conditions, and the crystals consist of a multitude of minute blocks. After 20 min annealing at 218°C the distance between some of the reflected maxima increases, while another group of maxima remains unchanged. After annealing times of 30 - 60 min the samples again show the same picture as immediately after quenching. Consequently, after the coherent bond between the newly formed material and the matrix has broken, the disorientated blocks return to their initial position. However, this elastic disorientation has a local nature and covers the total crystal volume non-uniformly. The degree of inhomogeneity is determined by the substructure of the initial crystal of the solid solution. There are 6 figures and 4 references: 3 Soviet and 1 non-Soviet. The reference to the English-language publication reads as follows: A. Guinier, J. Tennevin, Acta crystallogr. 2, 133 (1949).

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CIA-RDP86-00513R001963610011-3

ZAKHAROVA, M.I.(Moskva); MELIK-ADAMYAN, V.R.(Moskva) Investigating the substructure during the decomposition of solid sloutions of zinc in aluminum. Izv. AN SSSR.Otd.tekh.neuk. Met. 1 topl. no.5:210-211 S-0 '62. (MIRA 15:10) (Aluminum-zinc alloys-Metallography)

APPROVED FOR RELEASE: 09/19/2001

ACCESSION	PF-4/Pad/Ps-4 AS(mp)-2/ASD(m)-3/ JD/HH/ 8/0000/84/000/000/1318/0321. MLK
AUTHOR: Zakharoya, M.I. TITLE: Variation in the substructure of mets	is and alloys under thermomechanical
treatment SOURCE: AN SSSR. Nauchuy*y sovet po prof staley i splavov (Studies on steels and alloys) TOPIC TAGS: metal structure, alloy structure mechanical treatment, alloy hardening, zince martensitic transformation, nickel alloy 71 ABSTRACT: Thermomechanical treatment is of metals and alloys, particularly for increat article considers the resulting structural value	bieme zharoprochny*kh splavor. Issledovaniya Moscow, Izd-vo Nauka, 1964, 818-821 re, metal crystal, alloy orystal, thermo- aluminura, plastic deformation, 7 27 s widely used for improving the properties sing metal and alloy strength. The present riations for both alloys and pure metals.

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"APPROVED FOR RELEASE: 09/19/2001 CIA-RDP86-00513R001963610011-3

ACCESSION NR: AP4039253 AUTHORS: Zakharova, M. I.; Khatanova, N. A. TITLE: Investigation of the substructure of single crystals by the x ray focusing method SOURCE: Zavodskaya laboratoriya, v. 30, no. 6, 1964, 721-724 TOPIC TAGS: crystal substructure, x ray focusing, microblock, macroblock, gonio- metric measurement, aluminum alloy, angular disorientation, polycrystalline specimen, metal annealing, metal tempering, microscope UMV 100 ABSTRACT: The authors used the method of A. Guinier and I. Tennevin (Acta Crysta ABSTRACT: The authors used the method of a specimen of alloy during thermal of a specimen of a lose of the order 1-2 cm ² . The thickness was determined by the atomic number of the alloy-forming element. For Al this is 1-2 mm. Goniometric measurements were made on specimen gradient furnaces at a speed of 10 mm/sec. The results of experiments on a block		BR
AUTHORS: Zakharova, M. I.; Khatanova, N. A. TITLE: Investigation of the substructure of single crystals by the x ray focusing method SOURCE: Zavodskaya laboratoriya, v. 30, no. 6, 1964, 721-724 TOPIC TAGS: crystal substructure, x ray focusing, microblock, macroblock, gonio- metric measurement, aluminum alloy, angular disorientation, polycrystalline specimen, metal annealing, metal tempering, microscope UMV 100 ABSTRACT: The authors used the method of A. Guinier and I. Tennevin (Acta Crysta 2, 133, 1949) to study the disorientation of a specimen of alloy during thermal of mechanical treatment. They measured the angular disorientation of a block to an accuracy of 10 seconds. The specimens they used had cross sections of the order 1-2 cm ² . The thickness was determined by the atomic number of the alloy-forming For Al this is 1-2 mm. Goniometric measurements were made on specimen	ACCESSION NR: APLO39253	s/0032/64/030/006/0721/0724
SOURCE: Zavodskaya laboratoriya, v. 30, no. 6, 1964, 721-724 TOPIC TAGS: crystal substructure, x ray focusing, microblock, macroblock, gonio- metric measurement, aluminum alloy, angular disorientation, polycrystalline specimen, metal annealing, metal tempering, microscope UMV 100 ABSTRACT: The authors used the method of A. Guinier and I. Tennevin (Acta Crysta 2, 133, 1949) to study the disorientation of a specimen of alloy during thermal of mechanical treatment. They measured the angular disorientation of a block to an mechanical treatment. The specimens they used had cross sections of the order 1-2 cm ² . The thickness was determined by the atomic number of the alloy-forming 1-2 cm ² . The thickness was determined by the atomic number of the alloy-forming the latter through	AUTHORS: Zakharova, M. I.; Khatanova, N. A. TITLE: Investigation of the substructure o	• f single crystals by the x ray focusing
2, 133, 1949) to study the disordered the angular disorientation of a block to an mechanical treatment. They measured the angular disorientation of a block to an accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds. The specimens they used had cross sections of the order accuracy of 10 seconds.	SOURCE: Zavodskaya laboratoriya, v. 30, no TOPIC TAGS: crystal substructure, x ray fo metric measurement, aluminum alloy, angular specimen, metal annealing, metal tempering,	A. Guinier and I. Tennevin (Acta Crystal)
obtained if on the speed of 10 mm/sec. The results of exportance	2, 133, 1949) to study the distribution mechanical treatment. They measured the an accuracy of 10 seconds. The specimens the 1-2 cm ² . The thickness was determined by	ngular disorientation of a block to an y used had cross sections of the order the atomic number of the alloy-forming ric measurements were made on specimens

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6 days it was 2°56'. Orig. art. has: 3 figures and 4 formulas. ASSOCIATION: Moskovskiy gosudarstvenny*y universitet im. M. V. Lomonosova (Moscow SUBMITTED: 00 DATE A(Q: 18Jun64) SUB CODE: MM NO REF SOV: COO OTHER: 001	ACCESSION NR: AP4039253 of monocrystalline solid sol tation of the specimen tempe 218C for 24 hours the value 6 days it was 2°56'. Orig.			gular disorien- annealing at 2°20', and after
SUB CODE: MM NO REF SOV: COO OTHER: OOL	ASSOCIATION: Moskovskiy gos State University)	udarstvennyky univ	ersitet im. M. V. Lom	
	UB CODE: MM			
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				OTHER: 001

AUTHOR: Bykov, V. N.; Rudenko, V. A.; Zakharova, M. I. TITLE: The redistribution of dislocations in a molybdenum single crystal by annealing 20 27 16 SOURCE: Fizika metallov i metallovedeniye, v. 19, no. 1, 1965, 145-147 TOPIC TAGS: dislocation redistribution, subgrain boundary, molybdenum single crystal, vacuum furnace, slip plane, subgrain fragmentation, dislocation rosette, pickling pit, vacuum annealing, Littice defect ABSTRACT: A study has been made of the redistribution of dislocations and the for- mation of subgrain boundaries in the process of annealing a sample of monocrystal- line molybdenum produced by electron-beam smelting. The groups of dislocations are usually arranged in the form of a dislocation "rosette," under the influence of concentrated local plastic deformations. In cast metals, local plastic defor- mation can be produced by the presence of submicroscopic pores which develop during the metal-cooling period. Annealing of the mentioned samples at temperatures of 1,500 and 2,000C results in a redistribution of the dislocations. Some of the latter shift to the boundaries of the subgrains and are absorbed by them. Others Card 1/2	L 31853-65 ENT(m)/EFF(n)-2/T/EN ACCESSION NR: AP5004276		IJF(c) JD/ 9/001/0145/0147		
Appealing SOURCE: Fizika metallov i metallovedeniye, v. 19, no. 1, 1965, 145-147 TOPIC TAGS: dislocation redistribution, subgrain boundary, molybdenum single crystal, vacuum furnace, slip plane, subgrain fragmentation, dislocation rosette, pickling pit, vacuum annealing, lattice defect AESTRACT: A study has been made of the redistribution of dislocations and the for- mation of subgrain boundaries in the process of annealing a sample of monocrystal- line molybdenum produced by electron-beam smelting. The groups of dislocations are usually arranged in the form of a dislocation "rosette," under the influence of concentrated local plastic deformations. In cast metals, local plastic defor- mation can be produced by the presence of submicroscopic pores which develop during the metal-cooling period. Annealing of the mentioned samples at temperatures of 1,500 and 2,000C results in a redistribution of the dislocations. Some of the latter shift to the boundaries of the subgrains and are absorbed by them. Others	AUTHOR: Bykov, V. N.; Rudenko,	and anguares of the state of the		srystal by	
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	ABSTRACT: A study has been made mation of subgrain boundaries in line molybdenum produced by elec- are usually arranged in the form of concentrated local plastic do mation can be produced by the pu- the metal-cooling period. Anne-	the process of ann the process of ann tron-beam smelting. of a dislocation " formations. In cas resence of submicros aling of the mention adjet ribution of the	The groups of rosette," unde at metals, loca scopic pores wh ned samples at a dislocations,	f dislocations the influence l plastic defendence i.ch develop de temperatures of Some of the	; ce pr- uring f

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L 31853-65 ACCESSION NR: AP5004276 contribute to the formation of new dislocation boundaries within the subgrains. The interaction between the dislocation grids and walls located in different planes results in the fragmentation of the old subgrains into smaller blocs. The formation of new subgrain boundaries also reveals intermittent and staggered shifts of dislocations. Orig. art. has: 7 photomicrographs. ŧ ASSOCIATION: None 4 SUBMITTED: 02Apr64 ENCL: 00 SUB COIE: SS NO REF SOV: 002 OTHER: 001 Card 2/2

APPROVED FOR RELEASE: 09/19/2001

ZAKHAROVA, M.I.; TUMAN'YAN, Yu.A.

Determination of the mutual orientation of crystals in solid solutions of Ge in Al and precipitating germanium crystals. Vest. Mosk.un. Ser. 3# Fiz., astron. 20 no.4:50-55 Jl-Ag '65. (MIRA 18:12)

1. Kafedra fiziki kristallov Moskovskogo gosudarstvennogo universiteta, Submitted April 26, 1964.

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ACC NR. AP6015772 (A, N) bound to an AUTHOR: Zakharova, M.I.; Mogarycheva, I.B.; Khatanova, N.A. OK: Physics Department, Moscow State University, im M.V.Lomonoscv (Fizichestiy fakultet Moskovskogo gosudarstvennogo universiteta) TITLE: Investigation of the initial stages of decomposition of the solid solution in Al-CurAg and Cu-Be-Ag alloys /Report, Fifth All-Union Conference on Electron Microscopy held in Sumy 6-8 July 1965 SOURCE: AN SSSR. Izvestiya. Seriya fizicheskaya, v. 30, no. 5, 1966, 808-812 TOPIC TAGS: aluminum base alloy, copper base alloy, solid solution, thermal docomposi- tion, electron microscopy, electron diffraction, x ray diffraction ABSTRACT: The changes in structure occurring incident to thermal-aging decomposition of the supersaturated solid solutions in Al + 3 % Cu + 7% Ag and Cu + 1.6% Be + 1.9% of the supersaturated solid solutions in Al + 3 % Cu + 7% Ag and Cu + 1.6% Be + 1.9% of the supersaturated solid solutions in Al + 3 % Cu + 7% Ag and Cu + 1.6% Be + 1.9% of the supersaturated solid solutions in Al + 3 % Cu + 7% Ag and Cu + 1.6% Be + 1.9% of the supersaturated solid solutions in Al + 3 % Cu + 7% Ag and Cu + 1.6% Be + 1.9% of the supersaturated solid solutions in Al + 3 % Cu + 7% Ag and Cu + 1.6% Be + 1.9% of the results obtained for the filminum-base alloy. The decomposition of the devoted to the results obtained for the filminum-base alloy. The decomposition of the aluminum-base alloy was studied at aging temperatures of 130 and 218°C. The initial aluminum-base alloy was studied at aging temperatures of 130 and 218°C. The initial aluminum-base alloy was studied at aging temperatures of 130 and 218°C. The initial aluminum-base alloy was studied at aging temperatures of 130 and 218°C. The initial aluminum-base alloy was studied at the zone stage, which is most clearly evinced after of decomposition at 130°		
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Al-CulAg and Cu-Be-Ag alloys Anoporty held in Sumy 6-8 July 1965/ SOURCE: AN SSSR. Izvestiya. Seriya fizicheskaya, v. 30, no. 5, 1966, 808-812 TOPIC TAGS: aluminum base alloy, copper base alloy, solid solution, thermal decomposi- tion, electron microscopy, electron diffraction, x ray diffraction ABSTRACT: The changes in structure occurring incident to thermal-aging decomposition ABSTRACT: The changes in structure occurring incident to thermal-aging decomposition ABSTRACT: The changes are by weight) were studied by electron microscopy, electron Ag alloys (the percentages are by weight) were studied by electron microscopy, electron Ag alloys (the results obtained for the filuminum-base alloy. The decomposition of the devoted to the results obtained for the filuminum-base alloy. The decomposition of the aluminum-base alloy was studied at aging temperatures of 130 and 218°C. The initial aluminum-base alloy was studied at aging temperatures of an advected after	the solution in	ру
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L 36560-66 ACC NR: AP6015772 \mathcal{L} to 200 Å). The former are enriched in silver; the latter - in copper. The crystallographic and other procedures employed for estimating the zone dimensions are described. A table gives the values of the spherical and Guinier-Preston zone dimensions as estimated from the x-ray diffraction and electron microscopic data for specimens aged for 2 days at 130° and for 30 min, 5 hours and 15 hours at 218°; the agreement is generally satisfactory. The same thing is true of the identified θ' and γ' phases (the phases were identified by plotting the reciprocal lattice and θ -phase networks). The data for the ternary alloy are compared with the analogous data for the binary Al + 3% Cu alloy, obtained by Hardy and Hill (reference cited in Russian translation) and some significant differences are noted. The decomposition of the solid solution in the Cu + 1.6% Be + 1.9% Ag alloy was studied by similar techniques after 5 min, 30 min, 13 hours, and 30 hours isothermal annealing at 218°. The results for this alloy are given only briefly. The electron diffraction data indicate that after 30 hours annealing the structure of this alloy consists of the matrix, spherical zones, γ' and γ phases and silver crystals. The microhardness is increased from 80 kg/mm² after quenching to 200 kg/mm² after 30 hours anneal. Several micrographs and diffraction patterns are reproduced in the text. Orig. art. has: 4 figures and 1 table. SUB CODE: 11, 20/ SUBM DATE: 00/ ORIG REF: 002/ OTH REF: 001

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L 04291-67 ENT(m)/T/ENP(t)/ETI IJP(c) ACC NR: AP6018945 SO	JH/JD URCE CODE: UR/0126/6		
AUTHORS: Zakharova, M. I.; Tuman'yan, Yu.		38	
RG: Moscow State University im. M. V. Lon	onosov (Moskovskiy go	suniversitet) \mathcal{B}	
TTLE: Decomposition of solid solution in $\frac{1}{\sqrt{2}}$ OURCE: Fisika metallov i metallovedeniye,	1-Ag-Ge and Al-Cu-Ge -7 -7 -7 v. 21, no. 6, 1966,	alloys 868-672	
TOPIC TAGS: thermal aging, aluminum base a containing alloy, silver containing alloy, BSTRACT: The effect of germanium upon the investigated. Methods employed in the study crystals, oscillation and rotation, and char prepared of Al (99.996%), Cu (99.9%), Ag (99 collowing compositions: 1) Al10% (by weight) Al3% Cu1%Ge. Decomposition of the sec at 20, 100, 130, and 218C. It was establish of Ge in Al10% Ag alloy almost entirely su cones during natural aging. Addition of Ge ffect upon the formation of these mones and t 130 and 218C. Orig. art. has: 4 figures UB CODE: 11/ SUBM DATE: 08Jun65/	aging of Al-Ag and Al aging of Al-Ag and Al were x-ray analysis ages in hardness. The (%), and Ge (99.99%) (ht); Ag2% Ge; 2) olid solution was observed to Al-Gu alloys also accelerates the sepa (0RIG REF: 002/	sition I-Cu alloys has been of the rigid mono- s alloys were and had the Al4% Cu0.4% Ge; prved after aging of 2% (by weight) Gin'ye-Preston has a retarding	

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UTHOR:	Khatanova, N. A.; Zakharova, M. I.	
DRG: Mo	scow State University im. M. V. Lomonosov (Moskovskiy gosuniversit	et)
SOURCE:	Fizika metallov i metallovedeniye, v. 22, no. 1, 1966, 55–57	
- 3- 12 000 1	AGS : electron microscope, alloy phase diagram, aluminum base alloy UEMV electron microscope	
ABSTRAC of lamell solid solid the autor by analyz	CT: The aging of the supersaturated solid solution of Cu in Al involves ar Guinier-Preston (G. P.) zones during the pre-segregation stage; the ation of Ag in Al involves the formation of spherical G. P. zones. In this is investigate the process of the decomposition of an Al-3 wt. $\%$ Cu-7 with zing anomalous effects on the roentgenograms of immobile monocrystal an electronmicroscopic analysis of thin foils following their aging at 13 : the investigated specimens contain both lamellar and spherical G. P. 0 min of aging at 218°C the photographs made with the aid of an UEMV-	. % Ag alloy and by per- 0 and 218°C. zones. Fol- 100 electron
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